



**HAL**  
open science

# Strongly Modified Mechanical Properties and Phase Transition in AlPO 4 -17 Due to Insertion of Guest Species at High Pressure

Frederico Alabarse, Benoît Baptiste, Yoann Guarnelli, Boby Joseph, Julien Haines

► **To cite this version:**

Frederico Alabarse, Benoît Baptiste, Yoann Guarnelli, Boby Joseph, Julien Haines. Strongly Modified Mechanical Properties and Phase Transition in AlPO 4 -17 Due to Insertion of Guest Species at High Pressure. *Journal of Physical Chemistry C*, 2023, 127 (29), pp.14528-14533. 10.1021/acs.jpcc.3c03513 . hal-04182577

**HAL Id: hal-04182577**

<https://hal.umontpellier.fr/hal-04182577v1>

Submitted on 2 Nov 2023

**HAL** is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



Distributed under a Creative Commons Attribution - NonCommercial 4.0 International License

# Strongly Modified Mechanical Properties and Phase Transition in $\text{AlPO}_4\text{-17}$ Due to Insertion of Guest Species at High Pressure

Frederico G. Alabarse,\* Benoît Baptiste, Yoann Guarnelli, Bobby Joseph, and Julien Haines\*



Cite This: *J. Phys. Chem. C* 2023, 127, 14528–14533



Read Online

ACCESS |



Metrics & More

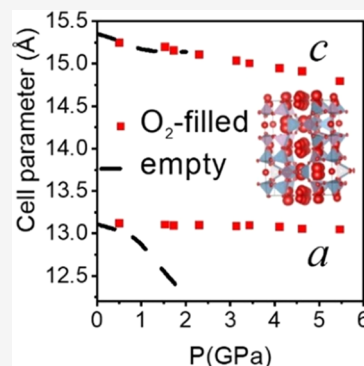


Article Recommendations



Supporting Information

**ABSTRACT:** The porous aluminophosphate  $\text{AlPO}_4\text{-17}$  with a hexagonal erionite structure, exhibiting very strong negative thermal expansion, anomalous compressibility, and pressure-induced amorphization, was studied at high pressure by single-crystal and powder X-ray diffraction in the penetrating pressure transmitting media  $\text{N}_2$ ,  $\text{O}_2$ , and Ar. Under pressure, these guest species were confirmed to enter the pores of  $\text{AlPO}_4\text{-17}$ , thus completely modifying its behavior. Pressure-induced collapse in the  $xy$  plane of  $\text{AlPO}_4\text{-17}$  no longer occurred, and this plane exhibited close to zero area compressibility. Pressure-induced amorphization was also suppressed as the elastic instability in the  $xy$  plane was removed. Crystal structure refinements at a pressure of 5.5 GPa indicate that up to 28 guest molecules are inserted per unit cell and that this insertion is responsible for the reduced compressibility observed at high pressure. A phase transition to a new hexagonal structure with cell doubling along the  $a$  direction was observed above 4.4 GPa in fluid  $\text{O}_2$ .



## 1. INTRODUCTION

The porous aluminophosphate  $\text{AlPO}_4\text{-17}$ , which is isostructural with the zeolite erionite (space group  $P63/m$ ), exhibits the highest negative thermal expansion<sup>1,2</sup> in the zeolite class of materials. The erionite structure of  $\text{AlPO}_4\text{-17}$  (Figure 1) is characterized by 4-, 6-, 8-, and 12-membered rings (MR) of tetrahedra forming large erionite cages (*eri*), with a diameter of 6.556 Å, and columns of smaller cancrinite cages (*can*) linked to double six-membered rings (D6MR) along  $z$ . The negative thermal expansion in this material is linked to the thermally excited transverse motion of oxygen atoms bridging the  $\text{AlO}_4$  and  $\text{PO}_4$  tetrahedra, which bring the Al and P atoms closer together.<sup>2</sup>

Atoms or molecules with a maximum kinetic diameter of about 3.4 Å can diffuse into the porosity of the  $\text{AlPO}_4\text{-17}$  structure via the 8-membered rings. The negative thermal expansion in  $\text{AlPO}_4\text{-17}$  was found to be tuned by the insertion of guest molecules at high pressure giving close to zero thermal expansion in the  $xy$  plane.<sup>3</sup>

In addition to negative thermal expansion,  $\text{AlPO}_4\text{-17}$  also exhibits uncommon mechanical properties at high pressure. This material exhibits an elastic instability, becoming increasingly softer in the  $xy$  plane upon compression.<sup>4</sup> Similar behavior has also been observed in zeolites,<sup>5</sup> cyanides,<sup>6,7</sup> and metal–organic frameworks.<sup>8</sup> In the case of  $\text{AlPO}_4\text{-17}$ ,<sup>4</sup> this instability gives rise to complete amorphization of the material at modest pressures of just above 2 GPa, which is lower than that observed for many zeolites<sup>9–20</sup> and aluminophosphates.<sup>21–23</sup> These mechanical properties could also be expected to be modified by pore filling. Thus, in the present study, nitrogen, oxygen, and argon with kinetic diameters<sup>24</sup> of

3.64, 3.46, and 3.40 Å, respectively, which are close to or slightly larger than the maximum size of 3.4 Å, were selected to be inserted in  $\text{AlPO}_4\text{-17}$  under high pressure in order to study guest insertion and to determine its effect on the compressibility and phase stability of this material.

## 2. EXPERIMENTAL METHODS

Single crystals of hydrated  $\text{AlPO}_4\text{-17}$  with maximum dimensions of  $250 \times 70 \times 70 \mu\text{m}^3$  were synthesized as described previously<sup>4,25</sup> from aluminum triisopropoxide and phosphoric acid using  $N,N,N',N'$ -tetramethyl-1,6-hexanediamine as a structure directing agent. The crystals were calcined in air at 500 °C for 24 h.

$\text{AlPO}_4\text{-17}$  single crystals (maximum dimensions  $190 \times 70 \times 70 \mu\text{m}^3$ ) or ground single crystals were placed in 235–300  $\mu\text{m}$  diameter and 80–110  $\mu\text{m}$  thick copper or stainless steel gaskets along with a ruby pressure calibrant in membrane diamond anvil cells with opening angles of between 50 and 100°. The DACs were placed in a cryogenic gas loading system, and the sample was dehydrated for 2 h at 110 °C under vacuum (4 Pa). Nitrogen, oxygen, or argon was then loaded cryogenically by condensing the corresponding gases.

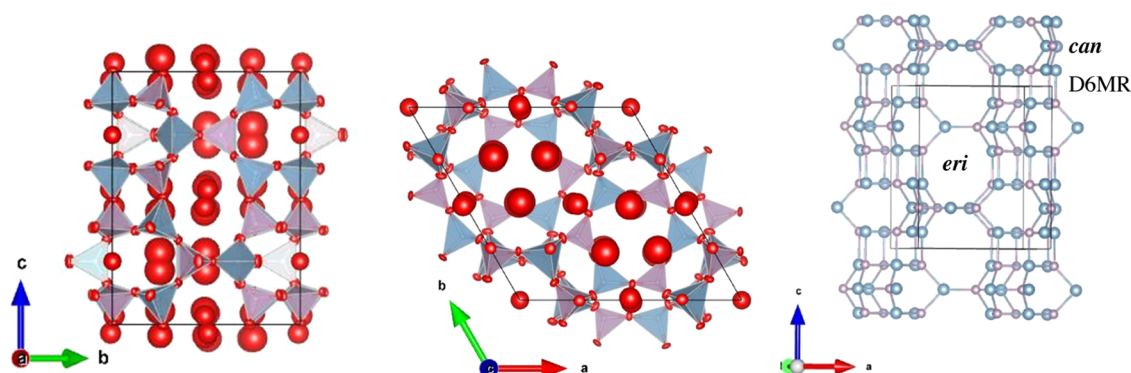
X-ray diffraction measurements ( $\lambda = 0.4957 \text{ \AA}$ ) under pressure were performed with an 80  $\mu\text{m}$  beam on the Xpress

Received: May 24, 2023

Revised: June 29, 2023

Published: July 18, 2023





**Figure 1.** Projections (left, center) of the crystal structure of  $\text{O}_2$ -filled  $\text{AlPO}_4$ -17 at 1.5 GPa. The tetrahedra correspond to the  $\text{AlO}_4$  and  $\text{PO}_4$  units of  $\text{AlPO}_4$ . Atomic displacement ellipsoids for  $\text{O}_2$  (red) are plotted at 50% probability. Note the larger atomic displacement for the oxygen atoms in the inserted  $\text{O}_2$  molecules as compared to those in the tetrahedra. Ring structures (*eri*, *can*, D6MR) are displayed (on the right) without the oxygen atoms.

beamline equipped with a PILATUS3 S 6 M (DECTRIS) detector at the Elettra Sincrotrone Trieste (Trieste, Italy). The detector was placed between 249 and 280 mm from the sample for the single-crystal measurements at 621–951 mm for the powder runs. The pressure was measured based on the shift in the  $R_1$  fluorescence line of ruby.<sup>26</sup>

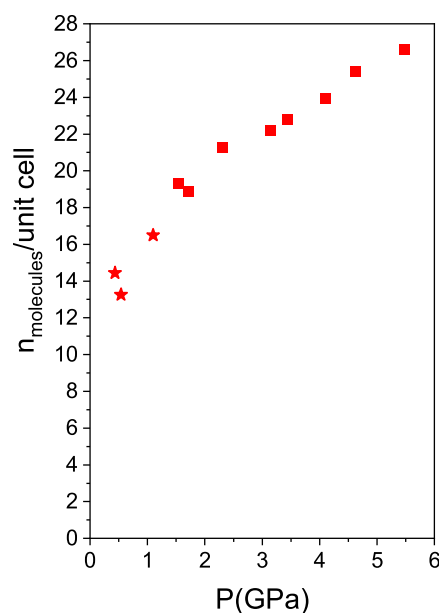
For the powder studies, the XRD images were converted to 1-D diffraction profiles using Dioptas.<sup>27</sup> Le Bail (Figures S1 and S2) and Rietveld refinements were performed using FullProf.<sup>28</sup> Le Bail and Rietveld refinements gave identical cell parameters; however, there are too many free atomic fractional coordinates to get accurate interatomic distances and angles for  $\text{AlPO}_4$ -17.

Diffraction data were collected from the  $\text{AlPO}_4$ -17 single crystals using phi scans from  $-30$  to  $+30^\circ$  or  $-45$  to  $+45^\circ$ , depending on the opening angle of the DAC. Data reduction was performed with CrysAlisPro 1.171.39.46 (Rigaku OD, 2018). The crystal structure was refined using Shelxl-2017/1<sup>29</sup> with the WinGX<sup>30</sup> and OLEX<sup>31</sup> interfaces. The Squeeze method<sup>32</sup> was applied using the OLEX interface. Crystal structures were plotted using Vesta.<sup>33</sup>

### 3. RESULTS AND DISCUSSION

Synchrotron X-ray diffraction studies were performed on both single crystals and powders of  $\text{AlPO}_4$ -17 in  $\text{N}_2$ ,  $\text{O}_2$ , and Ar. In the pressure range studied up to 6 GPa, good single-crystal XRD data were limited to  $\text{AlPO}_4$ -17 in  $\text{O}_2$  due to its higher solidification point of 5.5 GPa. Solidification of  $\text{N}_2$  and Ar occurs at much lower pressures of 2.4 and 1.5 GPa, respectively. Insertion of  $\text{O}_2$  in  $\text{AlPO}_4$ -17 was clear from Fourier difference maps (Figure S3). Refining the structures with guest atoms on fully occupied sites based on the peaks on the Fourier difference maps yielded R1 agreement factors in the range of 5.4–11.5%. This type of model, Figure 1, resulted in large atomic displacement parameters for the guest atoms as in ref 3, implying significant disorder, probably both positional and orientational, and corresponds to filling of the pores with 24 guest molecules per unit cell. In order to better quantify the degree of filling by the guest, the Squeeze method<sup>32</sup> was applied to determine the number of electrons and thus the number of guest molecules in the pores. This was found to be reliable only for a relatively large single-crystal  $125 \mu\text{m} \times 50 \mu\text{m} \times 50 \mu\text{m}^3$  in oxygen, for which the completeness was typically between 93 and 95% rather than 75% for other crystals investigated. This method resulted in a significant

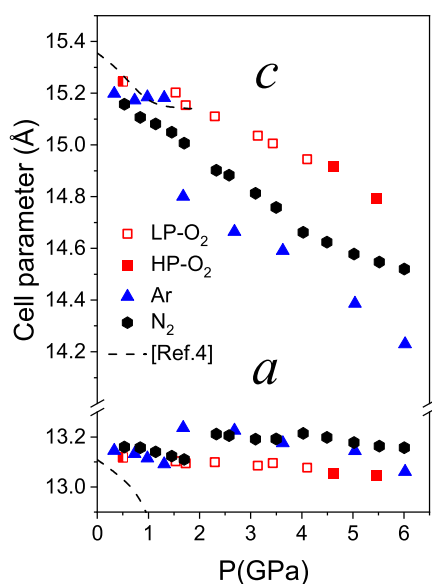
decrease in the R1 values to 2.8–4.3%. The obtained pore content per unit cell (Figure 2) was found to increase up to 28



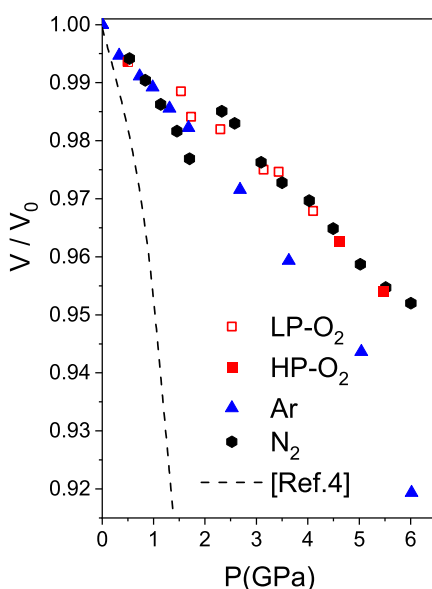
**Figure 2.** Guest content for oxygen-filled  $\text{AlPO}_4$ -17 as a function of pressure as obtained using the Squeeze method. The points at 0.43, 0.54, and 1.1 GPa (stars) were obtained on decompression.

molecules in the pressure range up to 4 GPa. This was slightly higher than the amount of  $\text{H}_2\text{O}$  molecules found in the large erionite cages of hydrated  $\text{AlPO}_4$ -17, for example (22  $\text{H}_2\text{O}$ /uc).<sup>25,26</sup>

This pore filling has major effects on the linear and volume compressibilities (Figures 3 and 4, Tables S1–S3) as compared to the intrinsic behavior of  $\text{AlPO}_4$ -17 in the nonpenetrating pressure transmitting medium, silicone oil.<sup>4</sup>  $\text{AlPO}_4$ -17 intrinsically presents an elastic instability, which manifests itself by an increasing collapse in the  $xy$  plane around the empty pores giving rise to a negative pressure derivative of the bulk modulus. Compression along  $z$  follows a normal behavior. It can be seen that the major effect of filling with guest atoms and molecules is to suppress the anomalous behavior in the  $xy$  plane due to filling of the pores. The  $a$  cell parameter initially is stable or slightly increases depending on the guest and then begins to compress slightly (see Figure 3). The  $c$  parameter



**Figure 3.** Unit cell parameters of guest-filled  $\text{AlPO}_4\text{-17}$  as a function of pressure. In the case of the high-pressure phase of  $\text{O}_2$ -filled  $\text{AlPO}_4\text{-17}$  (filled squares),  $a/2$  is plotted in order to directly compare the values to those of the low-pressure phase. Data for empty  $\text{AlPO}_4\text{-17}$  are from ref 4. Data at 0.5 GPa (half-filled square) for  $\text{AlPO}_4\text{-17}$  in  $\text{O}_2$  are from ref 3.

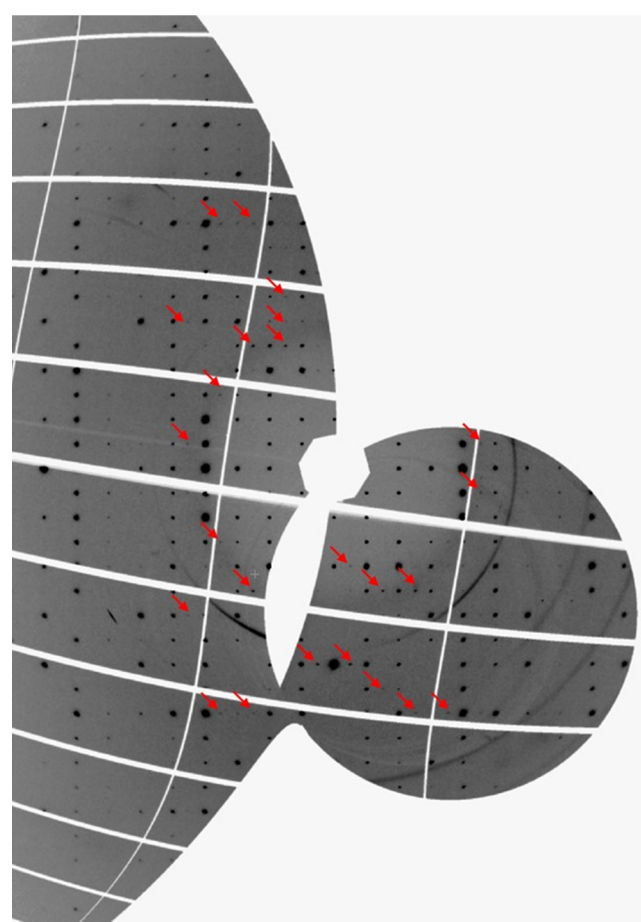


**Figure 4.** Relative volume of guest-filled  $\text{AlPO}_4\text{-17}$  as a function of pressure. Data for empty  $\text{AlPO}_4\text{-17}$  are from ref 4. The point at 0.5 GPa for  $\text{AlPO}_4\text{-17}$  in  $\text{O}_2$  is from ref 3.

decreases initially both in empty  $\text{AlPO}_4\text{-17}$  and in the filled systems. The behavior in argon appears to be strongly affected by the solidification of the fluid at 1.5 GPa. A very strong discontinuous decrease in  $c$ , along with a marked increase in  $a$ , occurs at this pressure, which can be linked to the appearance of nonhydrostatic stress. The effect of nonhydrostatic stress was also observed previously on compressing monoclinic, hydrated  $\text{AlPO}_4\text{-17}$  in  $\text{H}_2\text{O}$ <sup>34</sup> with discontinuities in the pressure dependence of the cell parameters and volume at the solidification pressure of  $\text{H}_2\text{O}$  at 0.9 GPa. In  $\text{N}_2$  in the fluid state, a discontinuous increase in  $a$  is observed, giving rise to increases in volume, Figure 4. In  $\text{N}_2$ , above the pressures at

which the volume increase occurs, and in  $\text{O}_2$ , compression principally occurs along  $c$ , with  $a$  being very stiff. This is similar to what has been observed as a function of temperature when  $\text{AlPO}_4\text{-17}$  is filled with  $\text{O}_2$ <sup>3</sup>, for which the thermal expansion along  $a$  is essentially zero, and strong negative thermal expansion is observed along  $c$ . The area compressibilities in the  $xy$  plane are  $1.2(2) \text{ TPa}^{-1}$  in  $\text{O}_2$  below the phase transition (see below) and  $1.8(4) \text{ TPa}^{-1}$  in  $\text{N}_2$  (above the volume jump) and are statistically zero over a range of at least 2 GPa. This essentially zero area compressibility is rare and is a phenomenon observed in relatively few materials, such as fluoroborates,<sup>35</sup> iodates,<sup>36</sup> and metal–organic frameworks.<sup>37</sup> The structural flexibility is restricted to the  $c$  direction. Upon decompression (Tables S1 and S2), the  $c$  lattice parameter increases very strongly, and  $a$  actually decreases. This behavior could be related to differences in the  $\text{O}_2$  content.

In the case of  $\text{AlPO}_4\text{-17}$  in  $\text{O}_2$ , which remains fluid up to 5.5 GPa, a phase transition occurs just above 4.1 GPa. A series of superlattice reflections of the type  $0\ 0.5\ 8$ ,  $0\ 0.5\ 5$ , and  $-0.5\ 1.5\ 5$ , for example, with a maximum relative intensity of 0.25%, appeared at the phase transition (Figure 5). This is consistent

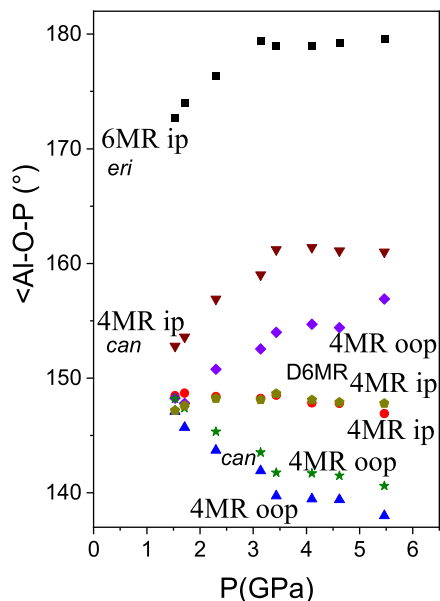


**Figure 5.**  $(0kl)$  Reciprocal space reconstruction for  $\text{AlPO}_4\text{-17}/\text{O}_2$  at 5.5 GPa. Arrows indicate the principal superlattice reflections.

with a doubling of the unit cell along  $a$ , while maintaining hexagonal symmetry. The very low intensity of the superlattice reflections would be compatible with a very slight distortion of the  $\text{AlPO}_4$  framework and/or eventually some slight degree of ordering of the  $\text{O}_2$  guest molecules; however, the number of observed superlattice reflections is too low to constrain such a

complicated structural model. A decrease in the  $a$  lattice parameter of 0.2% is observed at the phase transition, whereas  $c$  is less affected.

The low compressibility in the  $xy$  plane is linked to the opening of the in-plane Al-O-P angles in the 6MRs of the erionite cage and the stability of the in-plane Al-O-P angles in the 4MRs belonging to the D6MRs (Figures 6 and 7). The



**Figure 6.** In-plane (ip- $xy$ ) and out-of-plane (oop) Al-O-P angles in the 4MR and 6MR (see Figure 7) of the  $O_2$ -filled  $AlPO_4-17$  structure as a function of pressure.

diameter of the 6MRs of the erionite cage is very stable, changing from 2.5052(2) to 2.5025(4) Å from 1.5 to 5.5 GPa. In the case of the D6MRs, the diameter even expands slightly, varying from 2.454(4) to 2.488(5) Å over the same pressure range. The window defined by the 8MR expands in the  $xy$  plane from 3.31753(15) to 3.5468(5) Å and decreases along the  $c$  direction from 5.1353(3) to 4.9849(5) Å. There is a flattening of the cancrinite cages along  $c$  from 0.99696(15) to 0.6863(3) Å corresponding to a decrease in the out-of-plane Al-O-P angles in the 4MRs, decreasing the height of the 8MR

window. In contrast, the out-of-plane Al-O-P angles in the 4MRs belonging to the rigid double 6MRs between the cancrinite cages along  $c$  increase and open the 8MR window in the horizontal plane.

As in many other guest-filled porous materials,<sup>38–41</sup> in the filled  $AlPO_4-17$ , there is no tendency toward pressure-induced amorphization (PIA), which is essentially complete in the empty form just above 2 GPa. This can be a further indication that the mechanism of PIA is linked to the elastic instability arising from collapse in the  $xy$  plane, which is suppressed by filling the pores with guests. Molecular dynamics simulation shows that, similarly, the introduction of methane guests in the zeolitic imidazolate framework ZIF-8 increases the  $C_{44}$  elastic constant, the decrease in which is linked to the shear instability in the empty form at the origin of pressure-induced amorphization.<sup>42</sup>

#### 4. CONCLUSIONS

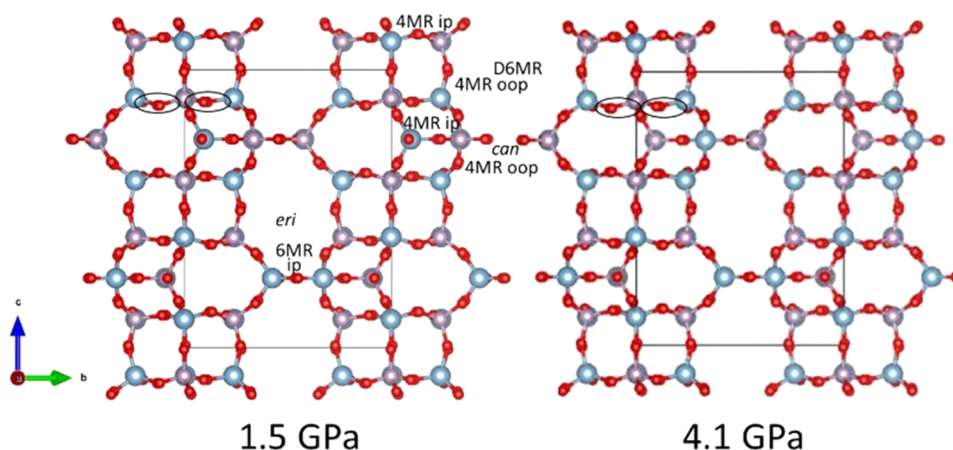
The present results show that argon, nitrogen, and oxygen readily enter the pores of  $AlPO_4-17$  at high pressure. This results in a pronounced change in the mechanical properties. Almost zero area compressibility is observed in the  $xy$  plane for  $AlPO_4-17$  in nitrogen and oxygen, and the elastic instability giving rise to pressure-induced amorphization is suppressed. A phase transition to a structure with a doubled unit cell is observed for  $AlPO_4-17$  in oxygen. The insertion of nonvolatile guest species could be a method for adapting the mechanical properties of such porous materials, in particular, compressibility and thermal expansion. This guest insertion could be used, for example, to adjust the thermal or elastic response of thin films of these materials to their supports in mechanical, electronic, or optical devices.

#### ■ ASSOCIATED CONTENT

##### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcc.3c03513>.

Unit cell parameters as a function of pressure (Tables S1–S3); additional crystallographic results (Figures S1–S3); and full single-crystal data for each pressure for  $AlPO_4-17$  in  $O_2$  (Figures S1–S3) (PDF)



**Figure 7.** Projections of the crystal structures of  $O_2$ -filled  $AlPO_4-17$  at 1.5 and 4.1 GPa indicating the in-plane (ip- $xy$ ) and out-of-plane (oop) Al-O-P angles in the 4MR and 6MR. The ellipses indicate the 4MR ip Al-O-P angles in the D6MR that vary by less than  $1^\circ$  over this pressure interval.

## AUTHOR INFORMATION

### Corresponding Authors

**Frederico G. Alabarse** – *Elettra Sincrotrone Trieste, Trieste 34149, Italy*; [orcid.org/0000-0002-7375-3666](https://orcid.org/0000-0002-7375-3666);  
Phone: +39 337 127 0679; Email: [frederico.alabarse@elettra.eu](mailto:frederico.alabarse@elettra.eu)

**Julien Haines** – *Institut Charles Gerhardt Montpellier, CNRS, Université de Montpellier, ENSCM, 34293 Montpellier, France*; [orcid.org/0000-0002-7030-3213](https://orcid.org/0000-0002-7030-3213); Phone: +33 448 79 21 14; Email: [julien.haines@umontpellier.fr](mailto:julien.haines@umontpellier.fr)

### Authors

**Benoit Baptiste** – *Institut de Minéralogie, de Physique des Matériaux et de Cosmochimie, (IMPMC), UMR 7590 CNRS—Sorbonne Université—IRD—MNHN, 75252 Paris, France*

**Yoann Guarnelli** – *Institut de Minéralogie, de Physique des Matériaux et de Cosmochimie, (IMPMC), UMR 7590 CNRS—Sorbonne Université—IRD—MNHN, 75252 Paris, France*

**Boby Joseph** – *Elettra Sincrotrone Trieste, Trieste 34149, Italy*; [orcid.org/0000-0002-3334-7540](https://orcid.org/0000-0002-3334-7540)

Complete contact information is available at:  
<https://pubs.acs.org/10.1021/acs.jpcc.3c03513>

### Author Contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

### Notes

The authors declare no competing financial interest.

## ACKNOWLEDGMENTS

The research leading to this result has been supported by the project CALIPSOplus under Grant Agreement 730872 from the EU Framework Programme for Research and Innovation HORIZON 2020. The synchrotron X-ray diffraction experiments were performed at the Xpress beamline from Elettra Sincrotrone Trieste (proposal numbers: 20200249 and 20215637). The authors acknowledge R. Borghes and V. Chenda for having improved the Xpress beamline software tools and for I. Cudin for the design and machining of the cryogenic loader.

## REFERENCES

- (1) Tschaufeser, P.; Parker, S. C. Thermal-Expansion Behavior of Zeolites and  $\text{AlPO}_4$ s. *J. Phys. Chem. A* **1995**, *99*, 10609–10615.
- (2) Atfield, M. P.; Sleight, A. W. Exceptional Negative Thermal Expansion in  $\text{AlPO}_4$ -17. *Chem. Mater.* **1998**, *10*, 2013–2019.
- (3) Alabarse, F.; Baptiste, B.; Joseph, B.; Haines, J. Tuning Negative Thermal Expansion in  $\text{AlPO}_4$ -17 by Insertion of Guest Molecules. *J. Phys. Chem. Lett.* **2022**, *13*, 9390–9395.
- (4) Alabarse, F. G.; Silly, G.; Brubach, J. B.; Roy, P.; Haidoux, A.; Levelut, C.; Bantignies, J. L.; Kohara, S.; Le Floch, S.; Cambon, O.; Haines, J. Anomalous Compressibility and Amorphization in  $\text{AlPO}_4$ -17, the Oxide with the Highest Negative Thermal Expansion. *J. Phys. Chem. C* **2017**, *121*, 6852–6863.
- (5) Fang, H.; Dove, M. T. Pressure-Induced Softening as a Common Feature of Framework Structures with Negative Thermal Expansion. *Phys. Rev. B* **2013**, *87*, No. 214109.
- (6) Chapman, K. W.; Chupas, P. J. Pressure Enhancement of Negative Thermal Expansion Behavior and Induced Framework Softening in Zinc Cyanide. *J. Am. Chem. Soc.* **2007**, *129*, 10090–10091.
- (7) Fang, H.; Phillips, A. E.; Dove, M. T.; Tucker, M. G.; Goodwin, A. L. Temperature-Dependent Pressure-Induced Softening in  $\text{Zn}(\text{CN})_2$ . *Phys. Rev. B* **2013**, *88*, No. 144103.
- (8) Chapman, K. W.; Halder, G. J.; Chupas, P. J. Pressure-Induced Amorphization and Porosity Modification in a Metal-Organic Framework. *J. Am. Chem. Soc.* **2009**, *131*, 17546–17547.
- (9) Liu, H. J.; Secco, R. A.; Huang, Y. N. Pressure-Induced Amorphization of Hydrated Na-X Zeolite. *PhysChemComm* **2001**, *4*, 37–39.
- (10) Gillet, P.; Malezieux, J. M.; Itie, J. P. Phase Changes and Amorphization of Zeolites at High Pressures: The Case of Scolecite and Mesolite. *Am. Mineral.* **1996**, *81*, 651–657.
- (11) Greaves, G. N.; Meneau, F.; Sapelkin, A.; Colyer, L. M.; Gwynn, I. A.; Wade, S.; Sankar, G. The Rheology of Collapsing Zeolites Amorphized by Temperature and Pressure. *Nat. Mater.* **2003**, *2*, 622–629.
- (12) Greaves, G. N.; Meneau, F. Probing the Dynamics of Instability in Zeolitic Materials. *J. Phys.: Condens. Matter* **2004**, *16*, S3459–S3472.
- (13) Greaves, G. N.; Meneau, F.; Kargl, F.; Ward, D.; Holliman, P.; Albergamo, F. Zeolite Collapse and Polyamorphism. *J. Phys.: Condens. Matter* **2007**, *19*, 415102–415117.
- (14) Colligan, M.; Forster, P. M.; Cheetham, A. K.; Lee, Y.; Vogt, T.; Hriljac, J. A. Synchrotron X-ray powder Diffraction and Computational Investigation of Purely Siliceous Zeolite Y Under Pressure. *J. Am. Chem. Soc.* **2004**, *126*, 12015–12022.
- (15) Isambert, A.; Angot, E.; Hebert, P.; Haines, J.; Levelut, C.; Le Parc, R.; Ohishi, Y.; Kohara, S.; Keen, D. A. Amorphization of Faujasite at High Pressure: an X-ray Diffraction and Raman Spectroscopy Study. *J. Mater. Chem.* **2008**, *18*, S746–S752.
- (16) Readman, J. E.; Forster, P. M.; Chapman, K. W.; Chupas, P. J.; Parise, J. B.; Hriljac, J. A. Pair Distribution Function Analysis of Pressure Treated Zeolite Na-A. *Chem. Commun.* **2009**, 3383–3385.
- (17) Haines, J.; Levelut, C.; Isambert, A.; Hebert, P.; Kohara, S.; Keen, D. A.; Hammouda, T.; Andrault, D. Topologically Ordered Amorphous Silica Obtained from the Collapsed Siliceous Zeolite, Silicalite-1-F: A Step toward “Perfect” Glasses. *J. Am. Chem. Soc.* **2009**, *131*, 12333–12338.
- (18) Vezzalini, G.; Arletti, R.; Quartieri, S. High-Pressure-Induced Structural Changes, Amorphization and Molecule Penetration in MFI Microporous Materials: A Review. *Acta Crystallogr., Sect. B: Struct. Sci., Cryst. Eng. Mater.* **2014**, *70*, 444–451.
- (19) Chan Hwang, G.; Joo Shin, T.; Blom, D. A.; Vogt, T.; Lee, Y. Pressure-Induced Amorphization of Small Pore Zeolites—the Role of Cation- $\text{H}_2\text{O}$  Topology and Anti-glass Formation. *Sci. Rep.* **2015**, *5*, No. 15056.
- (20) Thibaud, J. M.; Rouquette, J.; Hermet, P.; Dziubek, K.; Gorelli, F. A.; Santoro, M.; Garbarino, G.; Alabarse, F. G.; Cambon, O.; Di Renzo, F.; et al. High-Pressure Phase Transition, Pore Collapse, and Amorphization in the Siliceous 1D Zeolite, TON. *J. Phys. Chem. C* **2017**, *121*, 4283–4292.
- (21) Lv, H.; Yao, M. G.; Li, Q. J.; Liu, R.; Liu, B.; Lu, S. C.; Jiang, L. H.; Cui, W.; Liu, Z. D.; Liu, J.; et al. The Structural Stability of  $\text{AlPO}_4$ -5 Zeolite Under Pressure: Effect of the Pressure Transmission Medium. *J. Appl. Phys.* **2012**, *111*, No. 112615.
- (22) Alabarse, F. G.; Silly, G.; Haidoux, A.; Levelut, C.; Bourgogne, D.; Flank, A. M.; Lagarde, P.; Pereira, A. S.; Bantignies, J. L.; Cambon, O.; Haines, J. Effect of  $\text{H}_2\text{O}$  on the Pressure-Induced Amorphization of  $\text{AlPO}_4$ -54. *xH<sub>2</sub>O. J. Phys. Chem. C* **2014**, *118*, 3651–3663.
- (23) Alabarse, F. G.; Brubach, J. B.; Roy, P.; Haidoux, A.; Levelut, C.; Bantignies, J. L.; Cambon, O.; Haines, J.  $\text{AlPO}_4$ -54- $\text{AlPO}_4$ -8 Structural Phase Transition and Amorphization under High Pressure. *J. Phys. Chem. C* **2015**, *119*, 7771–7779.
- (24) Breck, D. W. *Zeolite Molecular Sieves: Structure, Chemistry and Use*; John Wiley & Sons, Inc: New York, 1974.
- (25) Tuel, A.; Lorentz, C.; Gramlich, V.; Baerlocher, C.  $\text{AlPO}_4$ -ERI, an Aluminophosphate with the ERI Framework Topology: Characterization and Structure of the As-Made and Calcined Rehydrated Forms. *C. R. Chim.* **2005**, *8*, 531–540.

(26) Shen, G.; Wang, Y.; Dewaele, A.; Wu, C.; Fratanduono, D. E.; Eggert, J.; Klotz, S.; Dziubek, K. F.; Loubeyre, P.; Fat'yanov, O. V.; et al. Toward an International Practical Pressure Scale: A Proposal for an IPPS Ruby Gauge (IPPS-Ruby2020). *High Pressure Res.* **2020**, *40*, 299–314.

(27) Prescher, C.; Prakapenka, V. B. DIOPTAS: a Program for Reduction of Two-Dimensional X-Ray Diffraction Data and Data Exploration. *High Pressure Res.* **2015**, *35*, 223–230.

(28) Rodriguez-Carvajal, J. Magnetic Structure Determination from Powder Diffraction using the Program FullProf. *Appl. Crystallogr.* **2001**, *30*–36.

(29) Sheldrick, G. M. Crystal Structure Refinement with SHELXL. *Acta Crystallogr., Sect. C: Struct. Chem.* **2015**, *71*, 3–8.

(30) Farrugia, L. J. WinGX and ORTEP for Windows: an Update. *J. Appl. Crystallogr.* **2012**, *45*, 849–854.

(31) Dolomanov, O. V.; Blake, A. J.; Champness, N. R.; Schroder, M. OLEX: New Software for Visualization and Analysis of Extended Crystal Structures. *J. Appl. Crystallogr.* **2003**, *36*, 1283–1284.

(32) Spek, A. L. PLATON SQUEEZE: A Tool for the Calculation of the Disordered Solvent Contribution to the Calculated Structure Factors. *Acta Crystallogr., Sect. C: Struct. Chem.* **2015**, *71*, 9–18.

(33) Momma, K.; Izumi, F. VESTA 3 for Three-Dimensional Visualization of Crystal, Volumetric and Morphology Data. *J. Appl. Crystallogr.* **2011**, *44*, 1272–1276.

(34) Alabarse, F. G.; Joseph, B.; Lausi, A.; Haines, J. Effect of H<sub>2</sub>O on the Pressure-Induced Amorphization of Hydrated AlPO<sub>4</sub>-17. *Molecules* **2019**, *24*, 2864.

(35) Jiang, X. X.; Luo, S. Y.; Kang, L.; Gong, P. F.; Yao, W. J.; Huang, H. W.; Li, W.; Huang, R. J.; Wang, W.; Li, Y. C.; et al. Isotropic Negative Area Compressibility over Large Pressure Range in Potassium Beryllium Fluoroborate and its Potential Applications in Deep Ultraviolet Region. *Adv. Mater.* **2015**, *27*, 4851–4857.

(36) Jiang, D.; Wen, T.; Song, H.; Jiang, Z.; Li, C.; Liu, K.; Yang, W.; Mao, H. K.; Wang, Y. Intrinsic Zero-Linear and Zero-Area Compressibilities over an Ultrawide Pressure Range within a Gear-Spring Structure. *CCS Chem.* **2022**, *4*, 3246–3253.

(37) Zeng, Q. X.; Wang, K.; Zou, B. Near Zero Area Compressibility in a Perovskite-Like Metal-Organic Frameworks [C(NH<sub>2</sub>)<sub>3</sub>][Cd(HCOO)<sub>3</sub>]. *ACS Appl. Mater. Interfaces* **2018**, *10*, 23481–23484.

(38) Haines, J.; Cambon, O.; Levelut, C.; Santoro, M.; Gorelli, F.; Garbarino, G. Deactivation of Pressure-Induced Amorphization in Silicalite SiO<sub>2</sub> by Insertion of Guest Species. *J. Am. Chem. Soc.* **2010**, *132*, 8860–8861.

(39) Graham, A. J.; Banu, A. M.; Duren, T.; Greenaway, A.; McKellar, S. C.; Mowat, J. P. S.; Ward, K.; Wright, P. A.; Moggach, S. A. Stabilization of Scandium Terephthalate MOFs against Reversible Amorphization and Structural Phase Transition by Guest Uptake at Extreme Pressure. *J. Am. Chem. Soc.* **2014**, *136*, 8606–8613.

(40) Thibaud, J. M.; Rouquette, J.; Dziubek, K.; Gorelli, F. A.; Santoro, M.; Garbarino, G.; Clement, S.; Cambon, O.; van der Lee, A.; Di Renzo, F.; et al. Saturation of the Siliceous Zeolite TON with Neon at High Pressure. *J. Phys. Chem. C* **2018**, *122*, 8455–8460.

(41) Poryvaev, A. S.; Polyukhov, D. M.; Fedin, M. V. Mitigation of Pressure-Induced Amorphization in Metal-Organic Framework ZIF-8 upon EPR Control. *ACS Appl. Mater. Interfaces* **2020**, *12*, 16655–16661.

(42) Ortiz, A. U.; Boutin, A.; Fuchs, A. H.; Coudert, F. X. Investigating the Pressure-Induced Amorphization of Zeolitic Imidazolate Framework ZIF-8: Mechanical Instability Due to Shear Mode Softening. *J. Phys. Chem. Lett.* **2013**, *4*, 1861–1865.