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## PAPER



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# Supramolecular co-assembly of water-soluble nucleobase-containing copolymers: bioinspired synthetic platforms towards new biomimetic materials†

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This study presents the development of co-assembled copolymer architectures at physiological pH (pH 7.4) formed *via* H-bonds between complementary nucleobase-containing copolymers. Well-defined hydrophilic copolymers were synthesised by RAFT polymerisation: statistical uracil- and thiomorpholine oxide-containing copolymers P(UrMA<sub>n</sub>-stat-THOXMA<sub>m</sub>) as well as diblock copolymers PEG<sub>112</sub>-b-P(AdMA<sub>n</sub>-stat-THOXMA<sub>m</sub>) composed of a PEG block and a second block of a copolymer of adenine- and thiomorpholine oxide-derived methacrylates. Binary mixtures of the resulting copolymers formed co-assembled nanoobjects in aqueous solution as a result of the H-bonds established between nucleobases. The influences of the polymer architecture (degree of polymerisation, co-monomer composition, length of the nucleobase-containing block), the ratio between complementary nucleobases, and the impact of H-bond competitors on the self-assembly properties were investigated. Light scattering techniques (SLS, DLS) and transmission electron microscopy (TEM) were used to characterise the co-assembled objects. This study demonstrates that the size of the resulting co-assemblies was mainly governed by the type and content of nucleobases, and by the length of the nucleobase block. Moreover, the *in vitro* evaluation of the nucleobase-containing polymers revealed that they were non-cytotoxic and hemocompatible. This study increases the understanding of nucleobase pairing in artificial copolymer architectures which are potential platforms for further use in biosciences.

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## Introduction

The structures of deoxyribonucleic acid (DNA) and ribonucleic acid (RNA) originate from self-assembly *via* H-bonds between complementary nucleobases such as adenine–thymine (in DNA) or adenine–uracil (in RNA).<sup>1</sup> In the case of DNA, the complementary nucleobases match perfectly to pack in a double helix structure.<sup>1</sup> RNA only has one strand, but could sometimes form secondary double helix structures as well.<sup>1</sup> While the self-assembly *via* nucleobase pairing is well known in the cases of DNA and RNA, it has been barely explored in

the case of synthetic copolymers containing nucleobases. Interesting attempts in this context were reported in the field of peptide nucleic acids (PNAs).<sup>2</sup> The structure of PNAs consists of peptide-like backbones derived from 2-aminoethylglycine containing sequences of nucleobases able to form various supramolecular architectures with helicoidal morphologies mimicking the DNA. Numerous nucleobase-containing polymers reported so far were prepared by post-functionalisation of synthetic polymers such as PEG (polyethylene glycol), PCL (polycaprolactone) or PPG (polypropylene glycol).<sup>3–5</sup> However, reaching high degrees of functionalization which would induce enhanced cooperativity by multiple H-bonds remains challenging. A solution is to polymerize vinyl monomers bearing nucleobases.

This strategy was implemented to prepare nucleobase-containing copolymers by Ring-Opening Metathesis Polymerisation (ROMP)<sup>6,7</sup> or Atom Transfer Radical Polymerisation (ATRP).<sup>8</sup> Reversible addition fragmentation chain transfer (RAFT) polymerisation is arguably one of the most versatile polymerization techniques and it has been applied advantageously to prepare polymers incorporating a

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high density of nucleobases able to form well-defined self-assembled architectures.<sup>9–14</sup>

The use of RAFT allows the preparation of polymers with controlled molar masses, contents of nucleobases, architectures and chain-end functionality.<sup>15</sup> Interactions between complementary nucleobase-containing polymers made by RAFT led to a variety of self-assembled architectures with tailored morphologies (spheres, vesicles, cylinders).<sup>9–14,16</sup> Nevertheless, the hydrophobicity of the nucleobase moiety required the use of an organic solvent (DMF, DMSO, dioxane or chloroform) for the formation of self-assembled objects.<sup>9–14,16,17</sup> For example, Kang *et al.* (2015)<sup>9</sup> studied the self-assembly of poly(2-(2-(thymine-1-yl)acetoxyl)ethyl methacrylate)-*b*-poly(methyl methacrylate) diblock copolymers in different mixtures of CHCl<sub>3</sub>/dioxane and reported various morphological transitions, from long flexible cylinders (at 50% vol. CHCl<sub>3</sub>) to short worm-like structures (at 12.5% vol. CHCl<sub>3</sub>).

A few studies have reported the formation of self-assembled objects based on nucleobase polymers in an organic solvent/water mixture, using the solvent switch method.<sup>10,14,16,17</sup> Representative examples by Hua *et al.* illustrated various morphological shapes such as spheres or cylinders obtained in DMF/water, by using block copolymers consisting of a hydrophilic poly(4-acryloyl morpholine) (PNAM) block and a hydrophobic poly(adenine propyl acrylamide) (PAAm) block.<sup>10</sup> Therein, the self-assembly was mainly driven (in terms of size and morphologies) by hydrophobic interactions caused by the adenine heterocycle. Only a few examples of systems that form stacks of multiple supramolecular units (such as urea-containing toluidine heterocycles, peptides and nucleobases) are suitable to self-assemble in water. Their self-assembly is driven by both hydrogen bonds and hydrophobic interactions and is therefore rarely at the thermodynamic equilibrium in water.<sup>18–20</sup> The hydrophobic interactions introduce a new parameter influencing the self-assembly that was poorly studied experimentally due to the synthetic difficulty of modulating the self-assembling moiety.<sup>21</sup>

The present study targets this problem and deals with hydrophilic nucleobase-containing copolymers able to co-assemble under physiological conditions. Polymethacrylate copolymers containing nucleobase (adenine or uracil) and hydrophilic thiomorpholine oxide were synthesized using the RAFT process. Various macromolecular structures were finely tailored in terms of degrees of polymerisation and contents of nucleobases. A binary mixture of uracil- and adenine-containing copolymers led to a wide range of co-assembled structures in water at physiological pH (pH 7.4). In contrast to previously described systems, H-bonds between complementary nucleobases rather than hydrophobic interactions were the main driving force of the assemblies. The influences of the number of nucleobases, the ratios between complementary nucleobases and the polymer architecture on the co-assembly were studied by SLS, DLS and TEM. In addition, *in vitro* investigations (cytotoxicity and compatibility assays on red blood cells) were carried out to test the potential of these nucleobase-containing copolymers for future biological applications.

## Experimental section

### Materials

Methacryloyl chloride (97% purity) was acquired from Fluka (Switzerland) and distilled (50 °C, 400 mbar) before use. 2-Bromoethanol (95% purity), adenine (Ad, 99% purity) and uracil (Ur, 98% purity) were purchased from Alfa Aesar (Germany). Thiomorpholine (98% purity) and 3-bromo-1-propanol (97% purity) were bought from Fluorochem (UK). Hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>) solution (30 wt%) was received from Carlo Erba (France). 4-Dimethylaminopyridine (DMAP, 96% purity), triethylamine (TEA, 99% purity), 2-cyano-2-propyl benzodithioate (CTA, 97% purity), poly(ethylene glycol) methyl ether (4-cyano-4-pentanoate dodecyl trithiocarbonate) (macro-CTA), disodium phosphate basic dodecahydrate (Na<sub>2</sub>HPO<sub>4</sub>·12H<sub>2</sub>O, 95% purity), potassium carbonate (K<sub>2</sub>CO<sub>3</sub>, 99.9% purity), sodium hydride (NaH, 90% purity) and deuterated solvents (deuterated chloroform, CDCl<sub>3</sub>, and hexadeuterodimethyl sulfoxide, DMSO-*d*<sub>6</sub>) were provided by Sigma Aldrich. 2,2'-Azobis(2-methylpropionitrile) (AIBN, 98% purity) was acquired from Sigma Aldrich (Germany) and recrystallised from methanol at 65 °C before use. Sodium chloride (NaCl) and citric acid monohydrate (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O, 99% purity) were obtained from VWR Chemical. Sodium bicarbonate (NaHCO<sub>3</sub>, 95% purity) was purchased from Fluka (France). Dimethylformamide (DMF, 99.8% purity) was acquired from Fisher Scientific (Belgium). HEPES (4-(2-hydroxyethyl)-1-piperazineethanesulfonic acid) buffered saline solution (30 mM) was acquired from PromoCell (Germany). Dry solvents (dichloromethane, CH<sub>2</sub>Cl<sub>2</sub>, and acetonitrile, CH<sub>3</sub>CN) were purified using a PureSolv Micro solvent purification system purchased from Sigma Aldrich (USA). The dialysis membranes used for the purification of polymers (Spectra/Por 7 Pre-treated RC Dialysis Tubing, MWCO = 1 kDa, diameter 24 mm, 4.6 mL cm<sup>-1</sup>) were bought from Krackeler Scientific, USA. 2 mM L-glutamine 100 U mL<sup>-1</sup> penicillin and 100 µg mL<sup>-1</sup> streptomycin solutions were acquired from Biochrom. 10% fetal calf serum was obtained from FCS, Capricorn Scientific. PrestoBlue solution was acquired from Thermo Fisher, Germany. Human blood was provided by the Department of Transfusion Medicine from Jena University Hospital (Germany). Branched poly(ethylene imine) (bPEI) solution was purchased from Polysciences Inc.

### Instrumentation

<sup>1</sup>H and <sup>13</sup>C-NMR spectra were recorded on a NMR Bruker Avance 400 MHz or NMR Bruker Avance III HD 400 MHz spectrometer using CDCl<sub>3</sub> or DMSO-*d*<sub>6</sub> as a deuterated solvent. The chemical shifts of protons were relative to tetramethylsilane (TMS) at δ = 0.

Size exclusion chromatography (SEC) was performed in DMF containing 0.1 wt% LiCl, with a flow rate of 0.8 mL min<sup>-1</sup> at 40 °C. Samples were filtered using TE36 Whatman PTFE-supported membrane filter paper (0.45 µm, 47 mm diameter) before the injection. The data were calibrated using polymethyl methacrylate (PMMA) standards.

$dn/dc$  values in water were determined using a differential refractometer and were estimated from the integrated refractive index (RI)-signal knowing the polymer concentration. Light scattering measurements were performed using an LS spectrometer (from LS Instruments, Switzerland) incorporating a goniometer based multi-angle static light scattering (SLS) and dynamic light scattering (DLS) instrument. Transmission electron microscopy (TEM) analyses were conducted on a JEOL 1200 EXII-120 kV instrument.

L929 cells used for cytotoxicity tests were incubated by using 96 well plates from VWR, Germany. The fluorescence measurements used to determine the cell viability were assessed using an Infinite M200 PRO microplate reader from Tecan, Germany. The haemoglobin release measurements and the cell aggregation rates were obtained using a plate reader from Tecan, Männedorf, Switzerland.

## Methods

### Synthesis of 3-bromopropyl methacrylate and 2-bromoethyl methacrylate

To a solution of 4-(dimethylamino)pyridine (DMAP) (162.4 mg, 0.05 eq., 16.8 mmol) in  $\text{CH}_2\text{Cl}_2$  (100 mL), 3-bromopropanol (2.4 mL, 1 eq., 26.54 mmol) and triethylamine (TEA) (7.4 mL, 2 eq., 53.2 mmol) were added under continuous stirring. Then, methacryloyl chloride (2.6 mL, 1 eq., 26.54 mmol) was added dropwise in an ice bath and under an inert ( $\text{N}_2$ ) atmosphere. The reaction mixture was kept at room temperature and under an inert atmosphere for 5 h. The resulting mixture was washed twice with a saturated  $\text{NaHCO}_3$  aqueous solution ( $2 \times 100$  mL) and then with distilled water (100 mL). The organic layer was collected, dried over magnesium sulfate and concentrated under vacuum to give a yellow oil (5 g, yield: 91%). 2-Bromoethyl methacrylate was obtained by a similar procedure to that of 3-bromopropyl methacrylate, following a published protocol.<sup>22</sup>

### Synthesis of 2-ethyl thiomorpholine oxide methacrylate (THOXMA)

2-Ethyl thiomorpholine oxide methacrylate was synthesized by oxidation of 2-ethyl thiomorpholine methacrylate, following a previously published procedure.<sup>23</sup>

### Synthesis of 3-(adenin-9-yl)propyl methacrylate (AdMA)

Adenine (1.5 g, 1 eq., 11.1 mmol) was dissolved in anhydrous DMF (100 mL) in a two-neck round-bottom flask and stirred for 1 h under an inert atmosphere at room temperature. Then, NaH (0.296 g, 1 eq., 11.1 mmol) was gently added and the mixture was kept under continuous stirring at room temperature for 30 min. Subsequently, freshly obtained 3-bromopropyl methacrylate (2.29 g, 1 eq., 11.1 mmol) was added and the flask containing the reaction mixture was immersed in an oil bath at 40 °C for 10 days. Then, the inorganic salts resulting from the reaction were removed by filtration. The resulting yellow filtrate was cryo-distilled to evaporate the organic solvent. A viscous yellow residue was formed after cryo-distilla-

tion. Anhydrous dichloromethane (100 mL) was added to the residue, and the resulting mixture was vigorously stirred for 30 min. Then, the mixture was filtered under vacuum and the filtrate was collected, dried with anhydrous magnesium sulphate and concentrated under vacuum to give a viscous yellow liquid (2 g, global yield: 68%). The monomer was characterised by  $^1\text{H}$  NMR (Fig. S1A†) and  $^{13}\text{C}$  NMR (Fig. S1B†) spectroscopy.  $^1\text{H}$  NMR, Fig. S1A† (400 MHz,  $\text{DMSO-d}_6$ )  $\delta$  (ppm) = 5.9 (d,  $\text{CH}_2$ , denoted as a); 5.58 (d,  $\text{CH}_2$ , denoted as a'); 1.87 (s,  $\text{CH}_3$ , denoted as b); 4.27 (t,  $\text{OCH}_2\text{CH}_2$ , denoted as c); 2.20 (m,  $\text{CH}_2\text{CH}_2\text{CH}_2$ , denoted as d); 4.10 (t,  $\text{NCH}_2\text{CH}_2$ , denoted as e); 7.26 (s,  $\text{NH}_2$ , adenine heterocycle, denoted as g); 8.15 (s,  $\text{N}=\text{CH}-\text{N}$ , adenine heterocycle, denoted as h); 8.13 (s,  $\text{N}=\text{CH}-\text{N}$ , adenine heterocycle, denoted as i).

### Synthesis of 3-(uracil-1-yl)propyl methacrylate (UrMA)

Uracil (1.5 g, 1 eq., 13.38 mmol) and anhydrous potassium carbonate (1.84 g, 1 eq., 13.38 mmol) were dissolved in anhydrous DMF (100 mL) in a two-neck round-bottom flask and stirred for 1 h under an inert atmosphere at room temperature. Then, 3-bromopropyl methacrylate (2.77 g, 1 eq., 13.38 mmol) was gently added using a syringe and the mixture was gently stirred at room temperature for 10 days. The mixture was filtered under vacuum and the filtrate was cryo-distilled to remove the organic solvent. After this step, dichloromethane (100 mL) was added and the resulting reaction mixture was stirred for another 15 min. Then, the mixture was filtered under vacuum to remove the inorganic salts. The filtrate was dried with anhydrous magnesium sulphate and then concentrated under vacuum, leading to a viscous pale yellow liquid (2 g, global yield: 63%). The product was characterised by  $^1\text{H}$  NMR (Fig. S2A†) and  $^{13}\text{C}$  NMR (Fig. S2B†) spectroscopy.  $^1\text{H}$  NMR, Fig. S2A† (400 MHz,  $\text{DMSO-d}_6$ )  $\delta$  (ppm) = 6.01 (d,  $\text{CH}_2$ , denoted as a); 5.58 (d,  $\text{CH}_2$ , denoted as a'); 1.87 (s,  $\text{CH}_3$ , denoted as b); 4.11 (t,  $\text{CH}_2\text{CH}_2$ , denoted as e); 1.96 (m,  $\text{CH}_2\text{CH}_2\text{CH}_2$ , denoted as d); 3.79 (t,  $\text{CH}_2\text{CH}_2$ , denoted as c); 7.62 (d,  $\text{CH}=\text{CH}$ , uracil heterocycle, denoted as g); 5.53 (d,  $\text{CH}=\text{CH}$ , uracil heterocycle, denoted as f); 11.5 (s,  $\text{NH}$ , uracil heterocycle, denoted as h).

### Synthesis of poly((3-(uracil-1-yl) propyl methacrylate)-*stat*-(2-ethyl thiomorpholine oxide methacrylate)) P(UrMA<sub>n</sub>-*stat*-THOXMA<sub>m</sub>) by RAFT polymerisation

In a typical protocol, 2-cyano-2-propyl benzodithioate (CTA, 1 eq.), AIBN (0.25 eq.), uracil methacrylate (UrMA) ( $x$  eq.) and thiomorpholine oxide methacrylate THOXMA ( $(y-x)$  eq. where  $y$  is the targeted DP) were dissolved in a mixture of DMF/aqueous phosphate buffer (pH 4,  $C_M = 4$  M) in a volume ratio (DMF/buffer) of 2 : 1. The mixture was thoroughly degassed *via* 3 freeze-pump-thaw cycles, filled with nitrogen and immersed in an oil bath at 80 °C. At different times, an aliquot of the reaction mixture was taken and analysed by  $^1\text{H}$ -NMR and SEC. After 7 h, the reaction was stopped by exposure to air. The mixture was then dialysed against water (with a 1 kDa MWCO membrane) for 3 days, followed by lyophilisation for 2 days. The resulting pink polymer powder was analysed by  $^1\text{H}$  NMR

in DMSO- $d_6$  and DMF SEC. For example, to obtain P(UrMA $_8$ -*stat*-THOXMA $_{34}$ ), the quantities of reagents used were CTA (1 eq., 8 mg, 0.036 mmol), AIBN (0.25 eq., 1.5 mg, 0.009 mmol), *uracil methacrylate* UrMA (10 eq., 82 mg, 0.36 mmol), and *thiomorpholine oxide methacrylate* THOXMA (40 eq., 332 mg, 1.44 mmol), dissolved in DMF (2 mL)/aqueous buffer (pH 4,  $C_M = 4$  M, 60 eq., 0.54 mL).  $^1\text{H}$  NMR, Fig. S3† (400 MHz, DMSO- $d_6$ )  $\delta$  (ppm) = 1.8 (s,  $\text{CH}_2$ , polymerizable synthon, denoted as **I<sub>b</sub>**); 1.8 (m,  $\text{CH}_2$ , UrMA aliphatic linker, denoted as **II<sub>b</sub>**); 3.43 (d,  $\text{CH}_2$ , UrMA aliphatic linker, denoted as **II<sub>a</sub>** and **II<sub>c</sub>**); 3.43 (d,  $\text{CH}_2$ , THOXMA aliphatic linker, denoted as **II<sub>d</sub>** and **II<sub>e</sub>**); 2.62 (t,  $\text{CH}_2\text{CH}_2$ , thiomorpholine oxide cycle, denoted as **III<sub>b</sub>**); 3.87 (t,  $\text{CH}_2\text{CH}_2$ , thiomorpholine oxide cycle, denoted as **III<sub>a</sub>**); 0.87, 0.97, 1.1 (s,  $\text{CH}_3$ , denoted as **I<sub>a</sub>**); 7.57 (d,  $\text{CH}=\text{CH}$ , uracil heterocycle, denoted as **i**); 5.87 (d,  $\text{CH}=\text{CH}$ , uracil heterocycle, denoted as **h**); 11.5 (s,  $\text{NH}$ , uracil heterocycle, denoted as **g**).

### Synthesis of poly(ethylene glycol)-*b*-poly((3-(adenine-9-yl)propyl methacrylate)-*stat*-(2-ethyl thiomorpholine oxide methacrylate)) PEG $_{112}$ -*b*-P(AdMA $_n$ -*stat*-THOXMA $_m$ ) by RAFT polymerisation

The PEG-macro chain transfer agent (macro-CTA PEG, 1 eq.), *adenine methacrylate* AdMA ( $x$  eq.) and *thiomorpholine oxide methacrylate* THOXMA ( $(y - x)$  eq. where  $y$  is the targeted DP) and AIBN (0.25 eq.) were dissolved in a mixture of DMF/aqueous phosphate buffer (pH 4,  $C_M = 4$  M) in a volume ratio (DMF/buffer) of 2 : 1. The mixture was degassed *via* 3 freeze-pump-thaw cycles, backfilled with nitrogen and then immersed in an oil bath at 80 °C for 7 hours. The conversion of the reaction was monitored each hour by  $^1\text{H}$  NMR and SEC. The reaction was then quenched by immersion in a liquid nitrogen bath and exposure to air. The final mixture was dialysed against water (with a 1 kDa MWCO membrane) for 3 days, followed by lyophilisation for 2 days. The resulting pale-yellow powder was analysed by  $^1\text{H}$ -NMR spectroscopy in DMSO- $d_6$  and DMF SEC. For exemplification in the case of PEG $_{112}$ -*b*-P(AdMA $_{30}$ -*stat*-THOXMA $_{70}$ ), the quantities of reagents used were macro-CTA PEG (1 eq., 20 mg, 0.0036 mmol), AIBN (0.25 eq., 0.15 mg,  $9 \times 10^{-4}$  mmol), *adenine methacrylate* AdMA (30 eq., 28.2 mg, 0.11 mmol), and *thiomorpholine oxide methacrylate* THOXMA (70 eq., 58.2 mg, 0.25 mmol), dissolved in DMF (1 mL)/aqueous buffer (pH 4,  $C_M = 4$  M, 130 eq., 0.12 mL).  $^1\text{H}$  NMR, Fig. S4† (400 MHz, DMSO- $d_6$ )  $\delta$  (ppm) = 1.35 (s,  $\text{CH}_2$ , polymerizable synthon, denoted as **I<sub>b</sub>**); 1.35 (m,  $\text{CH}_2$ , Ad aliphatic linker, denoted as **II<sub>b</sub>**); 3.47 (m,  $\text{CH}_2$ , PEG, denoted as **IV**); 3.96 (d,  $\text{CH}_2$ , AdMA aliphatic linker, denoted as **II<sub>a</sub>** and **II<sub>c</sub>**); 3.96 (d,  $\text{CH}_2$ , THOXMA aliphatic linker, denoted as **II<sub>d</sub>** and **II<sub>e</sub>**); 4.44 (t,  $\text{CH}_2\text{CH}_2$ , thiomorpholine oxide cycle, denoted as **III<sub>a</sub>** and **III<sub>b</sub>**); 0.9 (s,  $\text{CH}_3$ , denoted as **I<sub>a</sub>**); 8.56 (s,  $\text{NH}_2$ , adenine heterocycle, denoted as **g**); 8.97 (s,  $\text{N}=\text{CH}-\text{N}$ , adenine heterocycle, denoted as **h**); 9.6 (s,  $\text{N}=\text{CH}-\text{N}$ , adenine heterocycle, denoted as **i**).

### Preparation of starting polymer solutions and self-assembled formulations

The starting polymer solutions (denoted as P1, P3-uracil polymer solutions and P2, P4 adenine polymer solutions see

Table 2 and Table S1†) were prepared in HEPES buffer (pH 7.4) at a concentration of  $5 \text{ g L}^{-1}$  and stirred overnight at room temperature. Then, self-assembled formulations were obtained by slowly adding the uracil solution to the adenine solution, and the resulting mixture was then stirred for 2 days (Schemes 3 and S1†). For example, to prepare formulation A (P1 + P2, 1 mL), P1 solution (0.55 mL from a solution at a concentration of  $5 \text{ g L}^{-1}$ , in HEPES buffer, calculated as stated using eqn (S15)†) was gently added using a micropipette (15 min, 200 rpm) to P2 solution (0.45 mL from a solution at a concentration of  $5 \text{ g L}^{-1}$ , in HEPES buffer, calculated according to eqn (S16)†). Then, the solution mixture was stirred for 2 days (200 rpm, at room temperature). The solutions of starting polymers (P1, P2, P3, P4) were filtered through  $0.2 \mu\text{m}$  pore size Waters filters (USA) prior to performing SLS, DLS and TEM characterization studies. The absence of the concentration variation due to filtration was checked. Dilutions were made by adding the filtered solvent (through  $0.2 \mu\text{m}$  filters) and then stirring the solution for 5 min (Fig. S9 and S10†).

### Static light scattering

Static light scattering (SLS) measurements were performed using an LS spectrometer operating with a vertically polarized laser of wavelength  $\lambda = 660 \text{ nm}$ . All measurements (including dilutions) were done at room temperature (25 °C), collected from  $30^\circ$  to  $90^\circ$  with an interval of  $5^\circ$ , from  $90^\circ$  to  $110^\circ$  with an interval of  $10^\circ$ , and up to  $150^\circ$  with an interval of  $20^\circ$ . Prior to measurements, filtered toluene and filtered buffer (through a  $0.2 \mu\text{m}$  pore size Waters filter membrane) were used as a reference and solvent, respectively.

The Rayleigh ratio ( $R_\theta$ ) of the solution was determined following eqn (1).<sup>24,25</sup>

$$R_\theta = \frac{I_{\text{solution}}(\theta) - I_{\text{solvent}}(\theta)}{I_{\text{toluene}}(\theta)} \times \left( \frac{n_{\text{solvent}}}{n_{\text{toluene}}} \right)^2 \times R_{\text{toluene}} \quad (1)$$

where  $I_{\text{solution}}$ ,  $I_{\text{solvent}}$ , and  $I_{\text{toluene}}$  are the average intensities scattered by the solution, the solvent, and the reference (toluene) respectively, and  $n_{\text{solvent}} = 1.333$  (water) and  $n_{\text{toluene}} = 1.496$ , and  $R_{\text{toluene}} = 1.33 \times 10^{-5} \text{ cm}^{-1}$  is the Rayleigh ratio of toluene for wavelength  $\lambda = 660 \text{ nm}$ .

At a given concentration  $C$ ,  $R_\theta$  is related to the apparent weight average molar mass of the scatterers (or apparent molecular weight),  $M_a$ , and to the structure factor,  $S(q)$ , which depends on the scattering wave vector, as shown using eqn (2).<sup>24,25</sup> It is important to underline that  $M_a$  corresponds to the true molar mass ( $M_w$ ) only in very dilute solutions, where the interactions between the scatterers can be neglected.<sup>25,26</sup> At high concentrations, interactions cause  $M_a$  to differ strongly from  $M_w$ . For this reason, in order to accurately evaluate the true  $M_a$  of the self-assemblies, SLS measurements were performed for concentrations ranging between  $5 \text{ g L}^{-1}$  and  $1 \text{ g L}^{-1}$ . Consequently, the curve representing each  $M_a$  corresponding to each tested concentration was fitted, in order to determine the real  $M_w$  of the self-assembled objects as the intercept.

$$R_\theta = K \times C \times M_a \times S(q) \quad (2)$$

with  $C$  being the polymer concentration in  $\text{g L}^{-1}$  and  $K$  being a constant:

$$K = \frac{4\pi^2 n_{\text{solvent}}^2}{\lambda^4 N_{\text{a}}} \left( \frac{\partial n}{\partial C} \right)^2 \quad (3)$$

where  $N_{\text{a}}$  is Avogadro's number.

The aggregation number ( $N_{\text{agg}}$ ) was expressed according to eqn (4):<sup>24,25</sup>

$$N_{\text{agg}} = \frac{M_{\text{w}}}{M_{\text{u}}} \quad (4)$$

where  $N_{\text{agg}}$  is the aggregation number of particles,  $M_{\text{w}}$  is the weight average molar mass of particles, and  $M_{\text{u}}$  is the molar mass of the unimer.  $M_{\text{u}}$  values were determined by multiplying their  $M_{\text{n}}$  (determined by  $^1\text{H}$  NMR spectroscopy) by the corresponding  $M_{\text{w}}/M_{\text{n}}$  values determined by SEC. In the case of co-assembled structures, the aggregation number was evaluated using the same equation, with a modification of the calculation of  $M_{\text{u}}$ . For self-assembled structures,  $M_{\text{u}}$  was expressed as a sum of the weight fractions of each polymer multiplied by  $M_{\text{polymer}}$ , according to eqn (5).  $M_{\text{polymer}}$  was calculated by multiplying  $M_{\text{n}}$  (determined by  $^1\text{H}$  NMR spectroscopy) by the corresponding  $M_{\text{w}}/M_{\text{n}}$  values determined by SEC.

$$M_{\text{u}} = w_{\text{polymer1}} \times M_{\text{polymer1}} + w_{\text{polymer2}} \times M_{\text{polymer2}} \quad (5)$$

### Dynamic light scattering

The particle size ( $D_{\text{H}}$ , expressed as the hydrodynamic diameter of the particle) of co-assembled polymers was determined by dynamic light scattering (DLS). The DLS instrumentation consisted of the same LS spectrometer used for SLS experiments. DLS measurements were performed at 25 °C with a He-Ne 630 nm laser module, at a detection angle of 150°.

### Transmission electron microscopy

TEM analyses were performed using a JEOL 1400+ instrument equipped with a numerical camera, operating with an acceleration voltage of 120 kV at 25 °C. TEM samples were prepared by placing a drop (10.0  $\mu\text{L}$ ) of self-assembled polymer solution onto a carbon coated copper grid for 20 s, blotted with filter paper and then dried under ambient conditions. The experiments were performed at a concentration of 0.1% w/w.

### Cytotoxicity assays

Cytotoxic evaluation of nucleobase containing copolymers was assessed using the mouse fibroblast cell line L929 (400620, CLS), as recommended by ISO10993-5. L929 cells were cultured in Dulbecco's modified eagle's medium with 2 mM L-glutamine supplemented with 10% fetal calf serum, 100 U  $\text{mL}^{-1}$  penicillin, and 100  $\mu\text{g mL}^{-1}$  streptomycin at 37 °C under a humidified 5% (v/v)  $\text{CO}_2$  atmosphere. To this, the L929 cells were firstly seeded at  $10^3$  cells per mL ( $10^4$  cells per well) in a 96 well plate and then incubated for 24 h. No cells were seeded in the outer wells. The medium was changed to fresh cell culture medium 1 h prior to treatment. Cold nucleo-

base containing polymer solutions prepared in 20 mM HEPES (4-(2-hydroxyethyl)-1-piperazineethanesulfonic acid) were added to the cells at various concentrations (from 5 to 700  $\mu\text{g mL}^{-1}$ ), and then the plates were incubated for 24 h. The control cells were incubated with fresh culture medium containing the same amount of HEPES as the treated cells. Moreover, the medium was replaced by a mixture of fresh culture medium and the resazurin-based solution PrestoBlue. After incubation for another 45 min at 37 °C under a humidified 5% (v/v)  $\text{CO}_2$  atmosphere, the fluorescence was measured at  $\lambda_{\text{ex}} = 560 \text{ nm}/\lambda_{\text{em}} = 590 \text{ nm}$  with the gain set to optimal, with untreated cells on the same well plate serving as negative controls. The negative control was standardized as 0% of metabolism inhibition and referred to as 100% viability. Cell viability below 70% was considered to be an indication of cytotoxic behaviour. All experiments were performed in triplicate.

### Erythrocyte aggregation

Red blood cells from human blood were treated with nucleobase containing polymers at physiological pH (pH 7.4). Human blood was provided by the Department of Transfusion Medicine of the Jena University Hospital. Erythrocyte suspensions in PBS were prepared and mixed at a volume ratio of 1 : 1 with polymer solutions as described above. After incubation at 37 °C for 2 h, the erythrocyte aggregation was measured at 645 nm. As positive and negative controls, erythrocytes were treated with 50  $\mu\text{g mL}^{-1}$  25 kDa branched poly(ethylene imine) (bPEI) solution or PBS buffer at pH 7.4. The aggregation activity of the polymer samples was expressed as the aggregation rate (eqn (6)):

$$\text{Aggregation rate} = \frac{1}{A_{(\text{sample})}} \quad (6)$$

where,  $A_{(\text{sample})}$  is the mean absorbance of a given sample.

### Hemolytic activity

The release of haemoglobin from erythrocytes was assessed in order to evaluate the damaging properties of the red blood cell membrane in the presence of nucleobase-containing copolymers. The blood was centrifuged at 4500g for 5 min. Subsequently, the pellets were washed three times with PBS (pH 7.4) by centrifugation at 4500g for 5 min. Furthermore, the erythrocytes were suspended in PBS at pH 7.4 to mimic the physiological conditions of blood/cytoplasm. Cold nucleobase containing polymer solutions were made in 20 mM HEPES (4-(2-hydroxyethyl)-1-piperazineethanesulfonic acid) at a stock concentration of 1  $\text{mg mL}^{-1}$ . Then the solutions were mixed at a volume ratio of 1 : 1 with cold erythrocyte suspensions and were incubated at 37 °C for 1 h. Erythrocyte suspensions were centrifuged at 2400g for 5 min. The release of hemoglobin in the supernatant was determined at 544 nm. The absorbance was measured using a plate reader. Additionally, determinations were conducted with washed erythrocytes either lysed with 1% Triton X-100 or suspended in PBS at pH 7.4 as a refer-

ence. The hemolytic activity of the polymers was calculated according to the following equation (eqn (7)):

$$\text{Hemolysis}(\%) = \left( \frac{A_{(\text{sample})} - A_{(\text{PBS})}}{A_{(\text{Triton X-100})}} \right) \times 100 \quad (7)$$

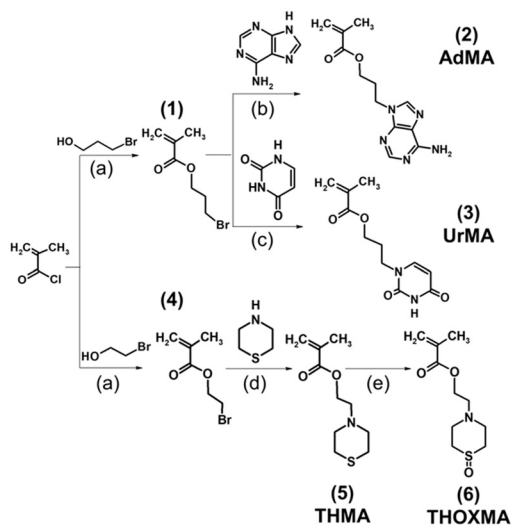
where  $A_{(\text{sample})}$ ,  $A_{(\text{PBS})}$ , and  $A_{(\text{Triton X-100})}$  are the absorbance of erythrocytes incubated with the polymer sample, suspended in PBS, and erythrocytes lysed with Triton X-100, respectively. This study was performed in line with the principles of the Declaration of Helsinki. All experiments were performed and approved by the ethics committee at the Friedrich Schiller University Jena (2021-2266-Material). Informed consent was obtained from human participants of this study.

## Results and discussion

### Nucleobase-containing methacrylates

The synthesis of adenine- and thymine-containing methacrylates has already been reported by Kang *et al.*<sup>13,22,27</sup> In this work, we adapted the previously reported synthetic routes to improve the yield and to reduce the synthetic efforts. Scheme 1 illustrates the synthesis of nucleobase-containing methacrylates used in the present study, called adenine methacrylate (AdMA) and uracil methacrylate (UrMA), which afforded the final monomer products with relatively high yields (above 63%) by straightforward techniques of organic chemistry.

These adenine- and uracil-methacrylates were insoluble in water. Our previous study reported the synthesis and characterisation of a new hydrosoluble thiomorpholine oxide methacrylate. Using RAFT polymerisation, hydrophilic polymers with



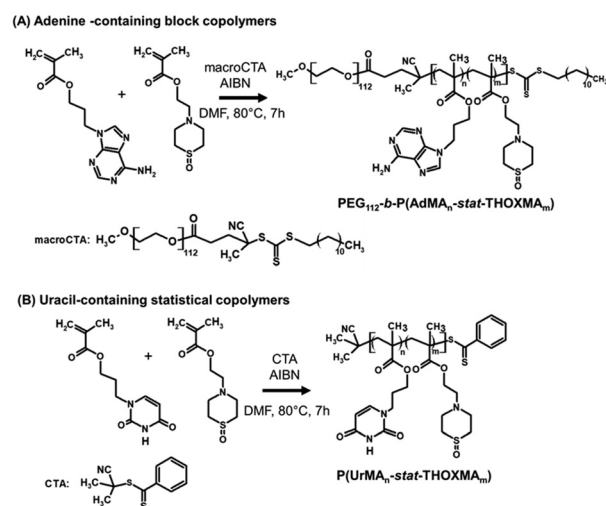
**Scheme 1** Syntheses of an adenine containing monomer (AdMA) (2), uracil methacrylate (UrMA) (3), thiomorpholine methacrylate (THMA) (5), and thiomorpholine oxide methacrylate (THOXMA) (6). (a): DMAP, TEA,  $\text{CH}_2\text{Cl}_2$ , RT, under  $\text{N}_2$ , overnight; (b): NaH, DMF, 40 °C, 10 days; (c):  $\text{K}_2\text{CO}_3$ , DMF, RT, 10 days; (d)  $\text{K}_2\text{CO}_3$ ,  $\text{CH}_3\text{CN}$ , under  $\text{N}_2$ , 40 °C, 6 days; (e):  $\text{H}_2\text{O}_2$ , under  $\text{N}_2$ , RT, overnight.

no cytotoxicity and high blood compatibility were synthesized.<sup>23</sup> Ethyl-thiomorpholine oxide methacrylate (THOXMA) was thus used as a hydrophilic comonomer for the synthesis of nucleobase-containing copolymers to modulate their hydrophobicity.

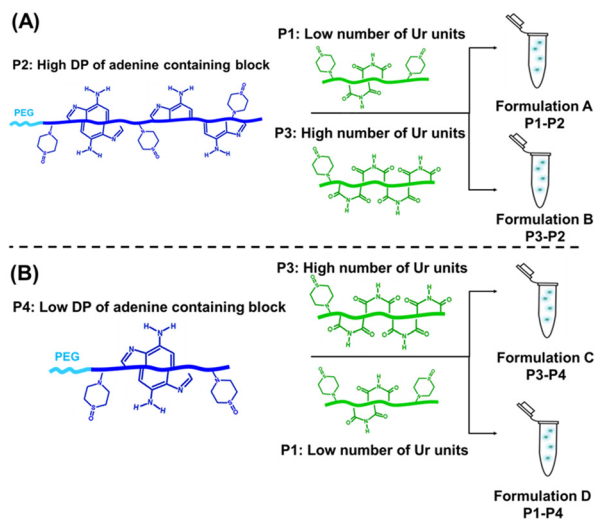
### Synthesis of nucleobase copolymers

Two families of nucleobase-containing copolymers were synthesised by RAFT polymerisation, as shown in Scheme 2: (A) diblock copolymers ( $\text{PEG}_{112}\text{-}b\text{-P}(\text{AdMA}_n\text{-}stat\text{-THOXMA}_m)$ ) composed of a hydrophilic block of PEG (with a constant DP = 112) and a second block driving the self-assembly prepared by the copolymerization of AdMA and THOXMA; (B) statistical copolymers of UrMA and THOXMA, denoted as  $\text{P}(\text{UrMA}_n\text{-}stat\text{-THOXMA}_m)$ . In all structures, the nucleobases (adenine, uracil) are H-bond promoters. In the case of the adenine-containing block copolymer,  $\text{PEG}_{112}\text{-}b\text{-P}(\text{AdMA}_n\text{-}stat\text{-THOXMA}_m)$ , the PEG block was selected for its high hydrosolubility and the steric stabilisation it can provide to self-assembled structures in water,<sup>28</sup> while THOXMA was introduced in the self-assembling moiety to modulate its hydrophobicity.

Both nucleobase copolymer families were synthesized by RAFT polymerization in a mixture of DMF/aqueous phosphate buffer (pH 4). The acidic buffer was used to prevent undesired aminolysis/hydrolysis of the dithioester and trithiocarbonate moieties of the chain transfer agents. The characterization data of these copolymers are summarized in Table 1. A good correlation between theoretical and experimental molar masses calculated by  $^1\text{H}$  NMR (Table 1) and comonomer ratios was observed (Table S3†). In addition, SEC analysis revealed a narrow molecular weight distribution ( $D$  ranging between 1.1 and 1.3, see Table 1) for all the copolymers, while the evolution of molar mass with conversion was linear, thus confirming that the polymerisation was controlled (Fig. S5 and S6†). These



**Scheme 2** Syntheses of nucleobase-containing copolymers: (A)  $\text{PEG}_{112}\text{-}b\text{-P}(\text{AdMA}_n\text{-}stat\text{-THOXMA}_m)$  diblock copolymers (P2, P4); (B)  $\text{P}(\text{UrMA}_n\text{-}stat\text{-THOXMA}_m)$  statistical copolymers (P1, P3).

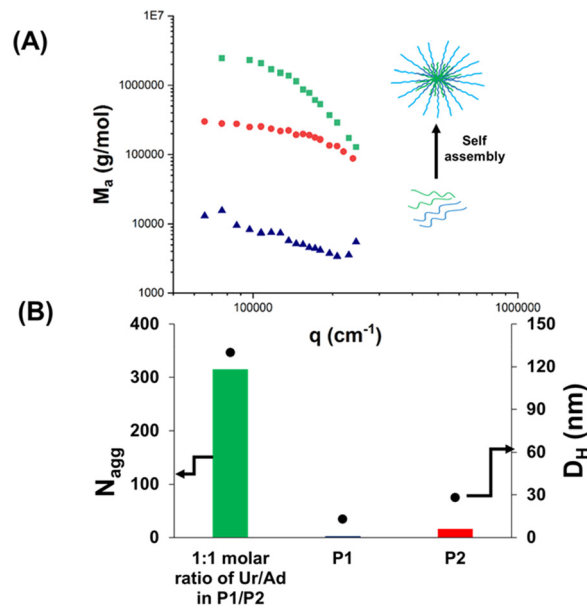


**Scheme 3** (A) Schematic representation of the composition of formulation A, made by adding a solution of P1 (with a low number of uracil units) to a solution of P2 (long DP of the adenine-containing block) and of the composition of formulation B, made by adding the solution of P3 (with a high number of uracil units) to a solution of P2 (long DP of the adenine-containing block); (B) schematic representation of the composition of formulation D, prepared by adding a solution of P1 (with a low number of uracil units) to a solution of P4 (short DP of the adenine-containing block) and of the composition of formulation C, formed by adding a solution of P3 (with a high number of uracil units) to a solution of P4 (short DP of the adenine-containing block).

results indicate that the RAFT polymerisation led to the formation of well-defined nucleobase-containing copolymers with controlled molar masses and desired compositions.

### Formation of co-assembled nucleobase polymer architectures at physiological pH

The obtained nucleobase copolymers were then dissolved in HEPES buffer (pH 7.4) and the properties of the resulting solutions were analysed (Scheme 3). First, starting filtered solutions of P1 and P2 were investigated by light scattering (SLS and DLS). The results are presented in Fig. 1, Table 2 and Table S2.† Low apparent molar mass ( $\sim 20\,000\text{ g mol}^{-1}$ ) and  $N_{\text{agg}}$  ( $\sim 1.6$ ) were observed for the uracil-containing copolymer (P1) (Table S2.†). This slight aggregation did not result in typical micellar structures observed in the case of previously



**Fig. 1** Co-assembly of P1 and P2 (formulation A): (A) apparent molar mass  $M_a$  evolution for individual polymer solutions (P1, P2) and formulation A, measured by SLS at a concentration of  $5\text{ g L}^{-1}$ ; (B)  $N_{\text{agg}}$  (evaluated by SLS) and  $D_H$  (evaluated by DLS) of individual polymer solutions and formulation A measured between  $1\text{ g L}^{-1}$  and  $5\text{ g L}^{-1}$ .

reported nucleobase blocks. The analysis of the adenine-containing copolymer (P2) showed a slightly higher aggregation number  $N_{\text{agg}}$  ( $\sim 7.5$ ) and apparent molar mass (Table S2.†). Solutions of P3 and P4 showed similar behaviour but slightly higher aggregation numbers (3.3 and 11.4, respectively) likely caused by their higher molar contents in hydrophobic nucleobases (Table 2 and Table S2.†). This slight aggregation is likely caused by hydrophobic interactions. Indeed the light scattering signatures of solutions of P1 or P2 were not affected by the addition of urea (Fig. 2).

Overall, in contrast to the pure hydrophobic nucleobase-containing polymers reported so far, these results show that uracil-(P1) and adenine-containing (P2) copolymers have no tendency to form large aggregates ( $N_{\text{agg}} < 8$  by SLS; particle hydrodynamic diameter  $D_H < 20\text{ nm}$ , by DLS, as shown in Table 2).

**Table 1** Characterisation of nucleobase-containing copolymers

Polymer name	Experimental DP <sup>a</sup>	$M_n$ ( $\text{g mol}^{-1}$ ), by <sup>1</sup> H-NMR <sup>b</sup>	$M_n$ ( $\text{g mol}^{-1}$ ), by SEC <sup>c</sup>	Dispersity ( $D$ ) <sup>c</sup>	Average number of nucleobases per polymer chain	$M_{\text{th}}$ ( $\text{g mol}^{-1}$ )	Theoretical target DP <sup>d</sup>
P1 P(UrMA <sub>8</sub> -stat-THOXMA <sub>34</sub> )	42	10 000	10 430	1.21	8	11 840	50
P2 PEG <sub>112</sub> -b-P(AdMA <sub>30</sub> -stat-THOXMA <sub>70</sub> )	104	30 200	32 100	1.32	30	29 400	100
P3 P(UrMA <sub>22</sub> -stat-THOXMA <sub>19</sub> )	41	10 500	11 200	1.1	22	11 950	50
P4 PEG <sub>112</sub> -b-P(AdMA <sub>5</sub> -stat-THOXMA <sub>5</sub> )	10	7900	9000	1.11	5	7700	15

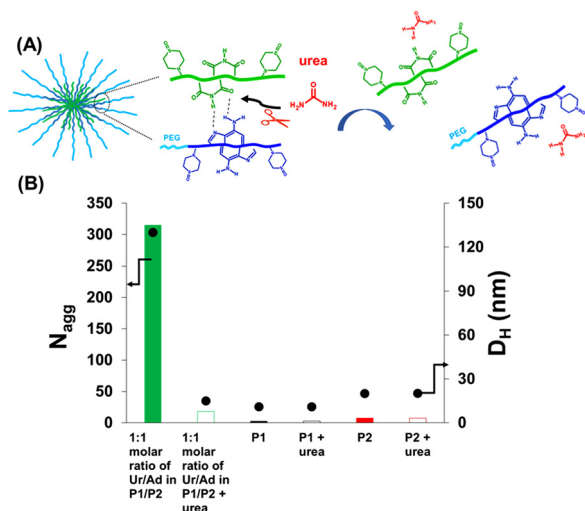
<sup>a</sup> Calculated by <sup>1</sup>H-NMR performed in DMSO-d<sub>6</sub>, according to eqn (S4) and (S9).† <sup>b</sup> Calculated by <sup>1</sup>H-NMR performed in DMSO-d<sub>6</sub>, according to eqn (S7) and (S14).† <sup>c</sup> SEC analysis performed in DMF containing 0.1% LiCl and by using PMMA standards. <sup>d</sup> Calculated using the following equation  $\text{DP}_{\text{target}} = ([\text{TTHOXMA}]/[\text{chain transfer agent}] \times \text{Conv}_{\text{TTHOXMA}}) + ([\text{AdMA or UrMA}]/[\text{chain transfer agent}] \times \text{Conv}_{\text{AdMA or UrMA}})$ .



**Table 2** Characterisation of co-assembled nucleobase-containing copolymer formulations

Formulation	Description	Aggregation number ( $N_{agg}$ ) <sup>a</sup>	Particle hydrodynamic diameter ( $D_H$ ) <sup>b</sup>
A	P1 + P2	315	130
B	P3 + P2	194	101
C	P3 + P4	9.95	40
D	P1 + P4	7.8	35
Polymer solution	P1	1.6	11
Polymer solution	P2	7.5	20
Polymer solution	P3	3.3	13.2
Polymer solution	P4	11.4	10.8

<sup>a</sup> Evaluated by SLS. <sup>b</sup> Evaluated by DLS; formulations A, B, C, and D were prepared with a Ur/Ad molar ratio of 1/1.



**Fig. 2** Disassembly of co-assembled nucleobase-containing copolymers P1 + P2 in HEPES buffer (pH = 7.4) in the presence of urea: (A) schematic representation of disassembly under urea influence; (B)  $N_{agg}$  (determined by SLS) and  $D_H$  (measured by DLS) of formulation A, P1 and P2 before vs. after the addition of urea measured between 1 g L<sup>-1</sup> and 5 g L<sup>-1</sup>.

Previous studies on DNA<sup>29</sup> indicated that the co-association between complementary nucleobases is stronger than the uracil–uracil association or adenine–adenine association. Thus, it is expected that the co-assembly between uracil- and adenine-containing copolymers led to self-assembled objects with high  $N_{agg}$  values and large particle sizes, if the complementary H-bond interactions are strong enough to overcome the competition with water molecules.

For this reason, the next step consisted in the investigation of the co-assembly of P1 and P2 in aqueous solution at physiological pH. Compared to individual P1 or P2 solutions, formulation A (P1 + P2, Ur/Ad ratio of 1/1) featured a high increase of the apparent molar mass  $M_a$  ( $\sim 10^7$  g mol<sup>-1</sup>) by SLS (Fig. 1A). This result proves the co-assembly of the complementary nucleobase-containing copolymers into aggregates with a high  $N_{agg}$  value ( $\sim 315$ ). The formation of co-assembled aggregates

was confirmed by DLS (Fig. 1B). Compared to P1 and P2, which showed low  $D_{He}$  values (particle hydrodynamic diameter < 20 nm), formulation A led to a  $D_H$  value 6.5 times higher (130 nm) than those of the starting polymers. These results suggest that the co-assembly originates from strong interactions between complementary nucleobases.

### Co-assembly in the presence of a H-bond competitor

Most of the self-assembled systems in water are driven by hydrophobic interactions. In order to prove unequivocally that the resulting co-assembly was mainly guided by H-bonds between complementary nucleobases instead of hydrophobic interactions, the effect of urea on the co-assembly was investigated. Urea is known as a strong competitor to H-bonds.<sup>30–33</sup> A diluted solution of urea (0.01 M) was added to formulation A (P1 + P2, Ur/Ad ratio of 1/1), and the mixture was stirred for 1 h and then examined by SLS and DLS. As presented in Fig. 2, a significant decrease of the aggregation number was observed after the addition of urea. Small particles with an average hydrodynamic diameter of 15 nm (measured by DLS) were formed. The size and the  $N_{agg}$  value were close to those of individual P1 and P2 copolymers (Fig. 1, 2 and Table 2). Urea cleaves the H-bonds between complementary nucleobases of P1 and P2 thus preventing the formation of co-assemblies, and leading to the release of P1- and P2-urea structures.<sup>30</sup> The dramatic effect of urea on formulation A (P1 + P2, Ur/Ad ratio of 1/1) confirms that H-bonds between complementary nucleobases are responsible for the co-assembly between P1 and P2.

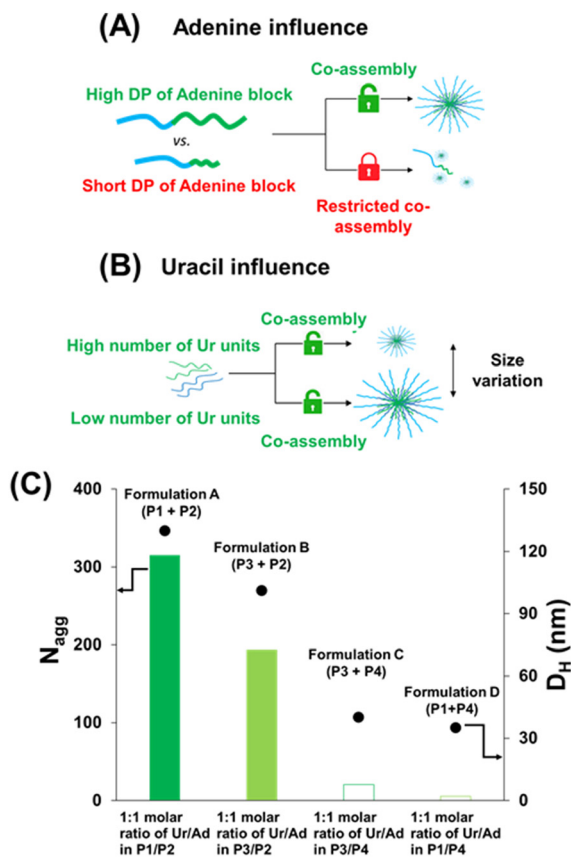
### Impact of adenine and uracil nucleobases

The effect of the length of the adenine-containing block and of the composition of the uracil containing polymer on the co-assembly was investigated. Four formulations (formulations A, B, C and D; Table 2 and Fig. 3) were compared. These formulations contained equimolar amounts of complementary nucleobases (ratio 1 : 1, eqn (S15) and (S16)<sup>†</sup>).

Formulations A and B were prepared using the same high length adenine-containing block copolymers (P2: PEG<sub>112</sub>-*b*-P(AdMA<sub>30</sub>-*stat*-THOXMA<sub>70</sub>), DP  $\sim 100$ , 30 units of adenine) and two different uracil containing copolymers (P1, P3) with the same degree of polymerisation (DP  $\sim 42$ ) but different numbers of uracil units (P1: 8 units of Ur and P3: 22 units of Ur).

SLS and DLS data of formulations A and B displayed in Fig. 3 show that aggregates with high  $N_{agg}$  values (above 200) and particle sizes (above 100 nm) were formed when a long adenine-containing block (DP  $\sim 100$ ) was used (Table 2 and Fig. 3).

However, significant differences were noted depending on the uracil copolymer (P1 or P3) used in the co-assembled formulations A and B. Formulation A (P1 + P2), prepared with P1 that contains only 8 uracil units per chain, led to objects with a high  $N_{agg}$  values ( $\sim 315$ , by SLS). In contrast, formulation B (P2 + P3), in which P3 has 22 uracil units per chain, produced smaller co-assemblies with  $N_{agg} \sim 194$  (by SLS, Fig. 3). DLS measurements revealed a similar trend for the hydrodynamic

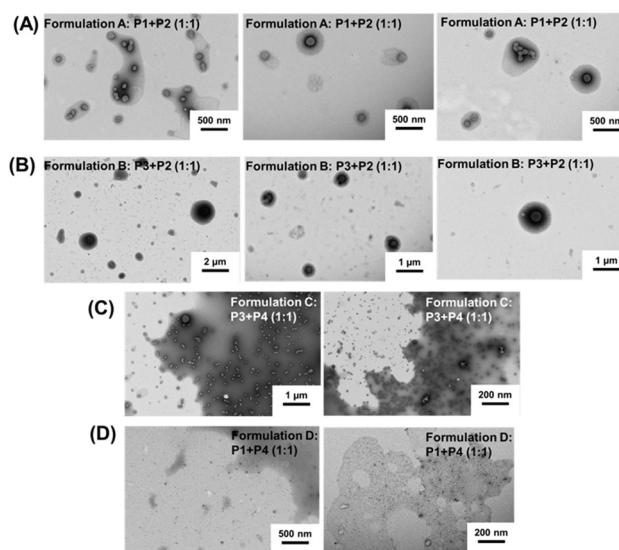


**Fig. 3** Impact of adenine (A) and uracil (B) nucleobases on the properties of co-assembled structures; (C) variation of  $N_{agg}$  by SLS and particle size ( $D_H$ ) by DLS of formulations A, B, C and D. The experiments were performed at concentrations between  $1 \text{ g L}^{-1}$  and  $5 \text{ g L}^{-1}$ .

particle diameters ( $D_H$ ) which reached 130 nm and 101 nm for formulation A and formulation B, respectively. Thus, increasing the number of uracil units per chain in the uracil-containing copolymer (from 8 to 22) led to smaller self-assembled objects. This might be because a smaller number of chains of the copolymer containing a higher density of uracil units (P3) is required to bind the complementary adenine containing copolymer (P2). This translates into a lower  $N_{agg}$  value and particle hydrodynamic diameter for formulation B than for formulation A. Spherical morphologies were observed by TEM in formulations A ( $\sim 150 \text{ nm}$ ) and B ( $\sim 100 \text{ nm}$ ) (Fig. 4), and their sizes were in agreement with the previous data obtained by DLS.

Formulations C and D were made using a shorter adenine containing block copolymer (P4: PEG<sub>112</sub>-*b*-P(AdMA<sub>5</sub>-stat-THOXMA<sub>5</sub>), DP  $\sim 10$ , 5 units of adenine) and the same uracil containing copolymers (P1, P3).

Low  $N_{agg}$  values and hydrodynamic diameters were determined when short length adenine blocks (formulations C and D) were used. For formulation C (using P3, DP = 42, 22 units of uracil), a very low  $N_{agg}$  value ( $\sim 10$ , by SLS) and particle size ( $\sim 40 \text{ nm}$ , by DLS) were observed. In addition, TEM analysis (Fig. 4) showed small spherical particles with a size of *ca.*



**Fig. 4** TEM characterisation of co-assemblies (A) formulation A: P1 + P2 mixture (1 : 1); (B) formulation B: P3 + P2 mixture (1 : 1); (C) formulation C: P3 + P4 mixture (1 : 1); (D) formulation D: P1 + P4 mixture (1 : 1). The co-assemblies were prepared at an initial concentration of  $1 \text{ g L}^{-1}$ . TEM experiments were performed at a concentration of  $0.1\% \text{ w/w}$ .

$35 \text{ nm}$  (formulation C, Fig. 4), in agreement with the DLS data. For formulation D (using P1, DP = 42, 8 units of Ur),  $N_{agg}$  ( $\sim 8$ ) was calculated by SLS (Fig. 3). This low  $N_{agg}$  ( $\sim 8$ ) obtained for formulation D in comparison to the results observed for formulation A shows that the use of polymers containing only small amounts and low density of adenine diminished the co-assembly ability to form large aggregates, likely because the H-bond cannot be formed in sufficiently high numbers.

Further investigation of the co-assembly in the presence of urea revealed a decrease of the particle hydrodynamic diameter and the number of aggregations in all formulations. The urea treatment leads to particles as small as  $30 \text{ nm}$  (Fig. S7†). This confirmed that the co-assemblies formed in the different formulations studied are driven by H-bonds between complementary nucleobase units rather than from hydrophobic interactions.

#### Influence of the molar ratios between Ur/Ad

Then, lower ratios of uracil and adenine equivalents ( $0.1/1$  molar ratio of Ur/Ad) were studied on four formulations (formulations E, F, G, H), as presented in Table S1.† Co-assembled aggregates (Table S1 and Fig. S8A†) with an average hydrodynamic diameter of around  $50 \text{ nm}$  (corresponding to single populations in DLS) were formed. These results suggest that at such low stoichiometries, the amount of uracil was likely too low to completely complex (by H-bonds) with adenine.

Moreover, high ratios between uracil and adenine equivalents ( $10/1$  molar ratio of Ur/Ad) were investigated by preparing additional formulations (formulations I, J, K, L, Table S1 and Fig. S8B†). Co-assembled formulations with a moderate  $N_{agg}$  ( $\sim 100$ ) and nanometric particle size ( $\sim 80 \text{ nm}$ ) were observed

when long adenine-containing blocks (formulations I and J) were used, while a low  $N_{\text{agg}}$  value (below 40) and particle size ( $\sim 30$  nm) were observed for the formulations based on short adenine-containing blocks (formulations K and L). These values were higher than those obtained for low ratios of Ur/Ad (*i.e.* Ur/Ad = 0.1/1), which is coherent since an increase in the number of complementary nucleobases increases the number of H-bonds which translates into the increase of  $N_{\text{agg}}$  and  $D_{\text{H}}$ . However, compared to the 1/1 stoichiometry (Fig. 3), the increase of the stoichiometry to 10/1 led to an overall decrease of  $N_{\text{agg}}$  and  $D_{\text{H}}$ . This behaviour was previously reported by Hua *et al.*<sup>10</sup> who explained this phenomenon to be a consequence of a low energy barrier for chain exchange which occurs at high stoichiometry, since the system is oversaturated in one of the complementary nucleobase copolymers. Smaller spherical aggregates were, therefore, formed to reduce the increased corona-chain repulsion introduced through the insertion of the complementary hydrophobic copolymer.<sup>10</sup> Applied to our system, the increase of the stoichiometry ratio increases the hydrophobicity of the system (by adding high amounts of the uracil containing copolymer) which led to more compact and smaller particles.

Globally, these results emphasize that structural parameters such as the copolymer architecture (DP, nucleobase number per polymer chain), the type of nucleobase, and the ratio between complementary nucleobases are important to control the size of the resulting co-assembled aggregates by H-bonds.

### Biocompatibility of nucleobase copolymers

Since the obtained nucleobase-containing copolymers were soluble at physiological pH and might be interesting for further biomedical applications, their *in vitro* biocompatibility was evaluated. Cytotoxicity, blood aggregation rates, and hemolytic activity evaluation are summarized in Fig. 5.

The cytotoxicity was assessed at pH 7.4 on the mouse fibroblast L929 cell line using the PrestoBlue assay (Fig. 5A).<sup>34</sup> This test works as a cell health indicator which analyses the reducing ability of living cells to finally assess cellular viability.<sup>34</sup> A cellular viability below 0.7 (or 70%) indicates a cytotoxic behavior.<sup>34</sup> As shown in Fig. 5A, the nucleobase containing copolymers showed a cellular viability above 95% up to the highest tested concentration of  $0.7 \text{ mg mL}^{-1}$ . Indeed, a slightly increased cell growth is observed for higher concentrations. A more detailed investigation of this phenomenon was however beyond the scope of this study.

These promising results led us to further evaluate the nucleobase containing copolymers towards blood compatibility, in terms of red blood cell aggregation activity (Fig. 5B) and hemoglobin release (Fig. 5C).

The aggregation activity was expressed by the aggregation rate and compared to a polycationic commercial polymer (*i.e.* polyethyleneimine PEI) which is known for its high aggregation rate (around 2.5).<sup>35</sup> All the tested copolymers showed an aggregation rate below 1.2 which underlines that they do not provoke undesired cell aggregation. This result could be explained by the neutral character of the nucleobase copoly-

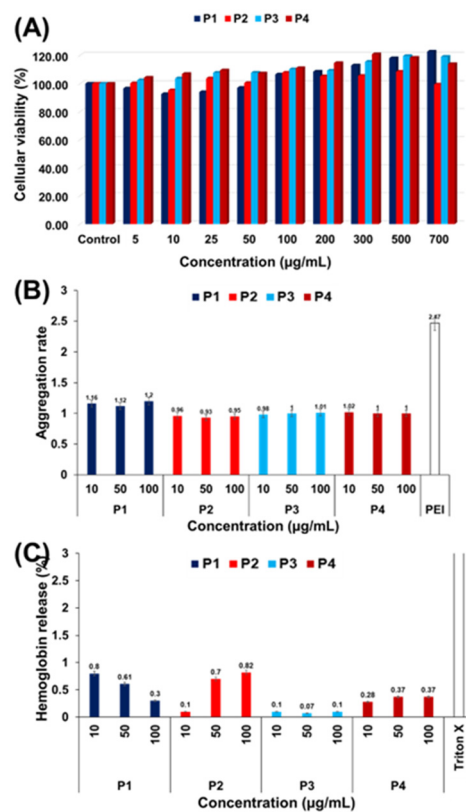


Fig. 5 Cytotoxicity of nucleobase containing copolymers evaluated on the L929 mouse cell line, pH 7.4, using non-treated cells as a control (A); blood aggregation rate of copolymers evaluated at pH 7.4, on human blood (B); hemoglobin release of nucleobase copolymers tested at pH 7.4 on human blood (C).

mers at physiological pH which avoided cellular aggregation, a frequent phenomenon observed for cationic polymers. Moreover, we investigated the interaction of nucleobase copolymers with the red blood cell membrane. In this regard, the release of hemoglobin at physiological pH (pH 7.4) from the erythrocytes was measured. As presented in Fig. 5C, the hemoglobin was released in low amounts (below 1%), which is consistent with the non-hemolytic activity.<sup>36</sup> Overall, these results emphasize promising potential of neutral and hydrosoluble nucleobase-containing copolymers as alternatives to cationic polymers in biological applications.

## Conclusions

A set of adenine- and uracil-containing statistical and block polymethacrylate copolymers have been successfully synthesized by RAFT polymerization. The resulting macromolecular structures were different from those previously reported in the literature, as a result of their hydrophilic character. The novelty of the developed co-assembled systems was that their preparation occurred exclusively in an aqueous buffer. The co-assembly was mainly driven by H-bonds between comp-

plementary adenine and uracil nucleobases, like in DNA. This approach is original compared to previously reported nucleobase polymethacrylate that self-assembled in organic solvents,<sup>5</sup> because of their insolubility in water. A range of structural parameters (number of nucleobase units per polymer, length of blocks and ratio between complementary nucleobases) were investigated to examine their influence on the co-assembly behaviour and the physicochemical properties (aggregation number, particle size) of the resulting nano-objects. Overall, spherical co-assembled morphologies were observed by TEM. However, significant modifications were observed (by SLS, DLS) due to the variations in the length of the nucleobase-containing block as well as in the content of complementary nucleobases. A high DP of adenine blocks led to self-assembled architectures with high  $N_{\text{agg}}$  values, while a low DP of adenine blocks enabled the formation of small-sized spherical co-assembled objects. The uracil block had a significant contribution to the regulation of the size of the co-assemblies. Low contents in uracil groups led to co-assembled objects with high aggregation numbers and particle sizes, whereas higher contents of uracil resulted in smaller particles. Structural parameters (such as the number of nucleobases per polymer and the length of the nucleobase block) could thus lead to different co-assembly behaviour. We proposed that these results are a consequence of possible dynamic H-bond interchanges between the nucleobases present in the copolymers. Furthermore, individual uracil or adenine polymers showed no ability to self-assemble. This observation emphasized that complementary adenine–uracil H-bond interactions were at the core of co-assembly formation. The co-assembly by H-bonds was proven by competitive assays with urea which led to a significant decrease of the  $N_{\text{agg}}$  value and particle size of the co-assembled objects. Evaluation of the *in vitro* biological properties of the nucleobase copolymers showed that they are not cytotoxic and compatible with blood. This work emphasized huge potential for the use of nature inspired H-bond interactions between complementary nucleobases to develop novel co-assembled architectures which may find future applications in biological systems.

## Author contributions

Conceptualization, S. Catrouillet and V. Ladmiral; methodology, L. V. Arsenie, S. Catrouillet, and V. Ladmiral; validation, S. Catrouillet and V. Ladmiral; formal analysis, L. V. Arsenie; investigation, L. V. Arsenie and M. Semsarilar; resources, V. Ladmiral, P. Lacroix-Desmazes and J. C. Brendel; writing—original draft preparation, L. V. Arsenie; writing—review and editing, L. V. Arsenie, M. Semsarilar, J. C. Brendel, P. Lacroix-Desmazes, V. Ladmiral, and S. Catrouillet; supervision, S. Catrouillet and V. Ladmiral; project administration, S. Catrouillet and V. Ladmiral; and funding acquisition, V. Ladmiral and P. Lacroix-Desmazes. All authors have read and agreed to the published version of the manuscript.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 F. Crick and J. Watson, *Nature*, 1953, **171**, 737–738.
- 2 C. Suparpprom and T. Vilaivan, *RSC Chem. Biol.*, 2022, **3**, 648–697.
- 3 C. W. Huang, W. Ji and S. W. Kuo, *Macromolecules*, 2017, **50**(18), 7091–7101.
- 4 I. Hong-Lin, C.-C. Cheng, C.-W. Huang, M.-C. Liang, J.-K. Chen, F.-H. Ko, C.-W. Chu, C.-F. Huang and F.-C. Chang, *RSC Adv.*, 2013, **3**, 12598–12603.
- 5 C.-C. Cheng, I.-H. Lin, J.-K. Chen, Z.-S. Liao, J.-J. Huang, D.-J. Lee and Z. Xin, *Macromol. Biosci.*, 2016, **16**(3), 1415–1421.
- 6 H. S. Bazzi and H. F. Sleiman, *Macromolecules*, 2002, **35**(26), 9617–9620.
- 7 H. Lu, J. Cai and K. Zhang, *Polym. Chem.*, 2021, **12**, 2193–2204.
- 8 M. Garcia, K. Kempe, D. M. Haddleton, A. Khan and A. Marsh, *Polym. Chem.*, 2015, **6**, 1944–1951.
- 9 Y. Kang, A. Pitto-Barry, H. Willcock, W.-D. Quan, N. Kirby, A. M. Sanchez and R. K. O'Reilly, *Polym. Chem.*, 2015, **6**, 106–117.
- 10 Z. Hua, A. Pitto-Barry, Y. Kang, N. Kirby, T. R. Wilks and R. K. O'Reilly, *Polym. Chem.*, 2016, **7**, 4254–4262.
- 11 Z. Hua, R. Keogh, Z. Li, T. R. Wilks, G. Chen and R. K. O'Reilly, *Macromolecules*, 2017, **50**(9), 3662–3670.
- 12 Z. Hua, T. R. Wilks, R. Keogh, G. Herwig, V. G. Stavros and R. K. O'Reilly, *Chem. Mater.*, 2018, **30**(4), 1408–1416.
- 13 Y. Kang, A. Lu, A. Ellington, M. C. Jewett and R. K. O'Reilly, *ACS Macro Lett.*, 2013, **2**(7), 581–586.
- 14 Y. Kang, A. Pitto-Barry, M. S. Rolph, Z. Hua, I. Hands-Portman, N. Kirby and R. K. O'Reilly, *Polym. Chem.*, 2016, **7**, 2836–2846.
- 15 H. Yang and W. Xi, *Polymers*, 2017, **9**(12), 666–690.
- 16 S. Varlas, Z. Hua, J. R. Jones, M. Thomas, J. C. Foster and R. K. O'Reilly, *Macromolecules*, 2020, **53**(22), 9747–9757.
- 17 G. B. Schuster, B. J. Cafferty, S. C. Karunakaran and N. V. Hud, *J. Am. Chem. Soc.*, 2021, **143**(25), 9279–9296.

- 18 S. C. Larnaudie, J. C. Brendel, I. Romero-Canélon, C. Sanchez-Cano, S. Catrouillet, J. Sanchis, J. P. C. Coverdale, J.-I. Song, A. Habtemariam, P. J. Sadler, K. A. Jolliffe and S. Perrier, *Biomacromolecules*, 2018, **19**(1), 239–247.
- 19 E. Obert, M. Bellot, L. Bouteiller, F. Andrioletti, C. Lehen-Ferrenbach and F. Boué, *J. Am. Chem. Soc.*, 2007, **129**, 15601–15605.
- 20 S. Han, E. Nicol, F. Niepceon, O. Colombani, S. Pensec and L. Bouteiller, *Macromol. Rapid Commun.*, 2018, 1800698–1800703.
- 21 D. Bochicchio, M. Salvalaglio and G. M. Pavan, *Nat. Commun.*, 2017, **147**(8), 1–11.
- 22 Y. Kang, A. Pitto-Barry, M. S. Rolph, Z. Hua, I. Hands-Portman, N. Kirby and R. K. O'Reilly, *Polym. Chem.*, 2016, **7**, 2836–2846.
- 23 L. V. Arsenie, F. Hausig, C. Kellner, J. C. Brendel, P. Lacroix-Desmazes, V. Ladmiral and S. Catrouillet, *Molecules*, 2022, **27**, 4233–4288.
- 24 K. E. B. Doncom, A. Pitto-Barry, H. Willcock, A. Lu, B. E. McKenzie, N. Kirby and R. K. O'Reilly, *Soft Matter*, 2015, **11**, 3666–3676.
- 25 J. P. Patterson, M. P. Robin, C. Chassenieux, O. Colombani and R. K. O'Reilly, *Chem. Soc. Rev.*, 2014, **43**, 2412–2425.
- 26 R. Xu, *Particuology*, 2015, **18**, 11–21.
- 27 H. A. Spijker, F. L. van Delft and J. C. M. van Hest, *Macromolecules*, 2007, **40**(1), 12–18.
- 28 J. Thevenot, A.-L. Troutier, L. David, T. Delair and C. Ladavière, *Biomacromolecules*, 2007, **8**(11), 3651–3660.
- 29 S. Scott, C. Shaheen, B. McGuinness, K. Metera, F. Kouzine, D. Levens, C. J. Benham and S. Leslie, *Nucleic Acids Res.*, 2019, **47**(12), 6360–6368.
- 30 W. Zhang, M. Liu, C. Lee, B. J. Salena and Y. Li, *Sci. Rep.*, 2018, **8**, 3–10.
- 31 W. K. Lim, J. Rosgen and S. Walter Englander, *Proc. Natl. Acad. Sci. U. S. A.*, 2009, **106**(8), 2595–2600.
- 32 A. Bende, *Theor. Chem. Acc.*, 2010, **125**, 253–268.
- 33 F. Jia, H. Li, R. Chen and K. Zhang, *Bioconjugate Chem.*, 2019, **30**(7), 1880–1888.
- 34 M. Xu, D. J. McCanna and J. G. Sivak, *J. Pharmacol. Toxicol. Methods*, 2015, **71**, 1–7.
- 35 A. Frère, M. Kawalec, S. Tempelaar, P. Peixoto, E. Hendrick, O. Peulen, B. Evrard, P. Dubois, L. Mespouille, D. Mottet and G. Piel, *Biomacromolecules*, 2015, **16**(3), 769–779.
- 36 I. Greco, N. Molchanova, E. Holmedal, H. Jensen, B. D. Hummel, J. L. Watts, J. Hakansson, P. R. Hansen and J. Svenson, *Sci. Rep.*, 2020, **10**, 1–13.