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The open-framework structure of KSbGe₃O₉ flux-grown crystals investigated by X-ray diffraction, vibrational spectroscopy, and DFT calculations.

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Abstract

We report on the preparation and the X-ray crystal structure of colorless KSbGe₃O₉, its vibrational properties (Raman and infrared studies) and density functional theory (DFT) calculations. KSbGe₃O₉, grown by the high-temperature flux method from K₂Mo₄O₁₃ flux, is thermally stable at least up to 1200°C and is isostructural to the benitoite BaTiSi₃O₉ (space group $P\overline{6}c2$ (n°. 188)). The hexagonal unit cell contains two formula units and the structure was refined to R1 = 0.0324 from single-crystal X-ray diffraction data. KSbGe₃O₉ is characterized with only one crystallographically independent Ge atom involved in three-member units [Ge₃O₉]⁶⁻ of regular germanate tetrahedra. The K⁺ ions are located in channels and, like Sb^V, are octahedrally surrounded by oxygen atoms. The KSbGe₃O₉ local structure and the planarity of Ge₃O₃ rings are also studied by a room-temperature vibrational investigation using non-polarized infrared and Raman spectroscopy. Both the infrared and Raman phonon modes have been assigned from the excellent agreement observed between our experimental data and the corresponding DFT ones. In particular, two E'(TO) modes (both IR and Raman active) characterize the planarity of the Ge₃O₃ ring in the *ab* plane.

KEYWORDS:

A.oxide materials, C.crystal structure, C.phonons, D. computer simulation, D. optical spectroscopy, D.X-ray diffraction

1. INTRODUCTION

Non-centrosymmetric germanate crystals with large nonlinear optical (NLO) properties are potential materials in the area of photonics and optical devices as they present electro-optical effects and second-order harmonic generation capabilities which able them to switch an optical signal and to generate a light with a frequency twice as the fundamental laser light [1-3]. Non-centrosymmetric germanate may also present interesting piezoelectric properties which are appropriate for volume- or surface-acoustic wave devices [4, 5]. In this context, the searching and the development of transparent and colorless germanate single-crystals with a non-centrosymmetric structure and enhanced properties are important. For industrial applications in harsh environment (with $T > 600^{\circ}$ C), lead-free materials that can operate under extreme stress without degradation should have a stoichiometric chemical composition and structural stability in a large thermal domain as encountered for the non-polar uniaxial α -GeO₂ single crystal [6, 7].

Based on the relationship between the tunnel structure and the physical properties of acentric germanate frameworks [8, 9], we have been interested in the AMGe₃O₉ family with a monovalent or bivalent (A = K, Rb, Ba, Ca, Tl) and a pentavalent or tetravalent (M = Sn, Nd, Ti, Ta) chemical element. These oxides which display the benitoite-type structure are characterized by a mixed framework built with MO₆ octahedra and GeO₄ tetrahedra, the latter forming a 3-member ring [Ge₃O₉]⁶⁻ with C_{3h} symmetry [10-15]. The GeO₄ units are known as microscopic NLO-active motifs and, when associated with a strongly electropositive alkaline element, the NLO efficiency of such oxide could be improved

[16]. Furthermore, these double germanate with an open-framework structure and well-defined channels may be suitable for ion-exchange and gas adsorption applications.

This paper reports on the spontaneous crystallization of the compound KSbGe₃O₉ using the hightemperature flux method and on its single crystal structure determination associated with a vibrational characterization. The experimental study is completed by a group theory analysis and density functional theory (DFT) calculations to investigate the KSbGe₃O₉ infrared (IR) and Raman spectra.

2. EXPERIMENTAL SECTION

Crystals were obtained using the flux method described elsewhere [17]. For the present study, $K_2Mo_4O_{13}$ was chosen as flux and synthesized via a solid-state reaction at 500°C during two weeks, from alkali carbonate K_2CO_3 and molybdenum oxide MoO₃ powders. The compound was characterized through classical methods and more precision on the conditions for EDX, spectroscopy or X-ray diffraction experiments can be found in our previous paper [17].

Calculations on KSbGe₃O₉ were performed within the density functional theory (DFT) framework as implemented in the ABINIT package [18]. The all-electron potentials are replaced by norm-conserving pseudopotentials generated according to the local density approximation [19]. K (4s¹), Sb (5s², 5p³), Ge (4s², 4p²) and O (2s², 2p⁴)-electrons are considered as valence states. The electronic wave functions are expanded in plane-waves up to a kinetic energy cutoff of 65 Ha and the integrals over the Brillouin zone were approximated by sums over an 8×8×6 mesh of special k-points according to the Monkhorst-Pack scheme [20]. The lattice parameters and the atomic positions were fully relaxed using a Broyden-Fletcher-Goldfarb-Shanno algorithm until the maximum stresses and residual forces were less than $2×10^{-4}$ GPa and $6×10^{-6}$ Ha/Bohr, respectively. Born effective charges, dielectric tensors, and dynamical matrix (yielding the phonon frequencies and eigenvectors) were obtained within a variational approach of the density functional perturbation theory, whereas the Raman susceptibilities were calculated from a nonlinear response formalism taking advantage of the 2*n*+1 theorem [21].

3. RESULTS AND DISCUSSION

3.1 Flux-growth result

Visually transparent and colorless as-grown crystals of almost millimeter size were obtained via spontaneous nucleation from the high-temperature flux method. Their chemical analysis shows the presence of only Ge, K and Sb in atomic ratios K/Sb of 1 and Ge/Sb of 3 which gives the nominal composition KSbGe₃O₉. DSC measurement from room temperature to 1200°C with a 10°/min heating rate did not point out any endo- or exo-thermal event. This means that the structure of KSbGe₃O₉ is thermally stable at least up to 1200°C, and that its melting temperature is over 1200°C.

Figure 1 shows the X-ray diffraction pattern θ -2 θ scan registered at room temperature for KSbGe₃O₉. The reflections can be indexed within the hexagonal $P\overline{6}c2$ space group (188) according to the JCPDS card number 00-038-1334 [22]. A small reflection attributed to the flux K₂Mo₄O₁₃ is present on the X-ray diffraction pattern (indicated by an arrow).

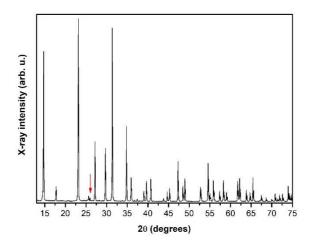


Figure 1: Powder X-ray diffraction pattern of KSbGe₃O₉ sample.

3.2. X-ray crystal structure.

The as-grown transparent platelets are not single-crystals as they look bright in any orientation under polarized light. Some attempts were made to carefully cut them into smaller single-crystals, but it rapidly became obvious, from their X-ray diffraction images, that such pieces were systematically twinned. However, two specimens were selected using a stereomicroscope equipped with a polarizing filter and the intensities diffracted by these crystals were collected. Because twinning occurs by very weak offset in the *ab* plane, a multi-crystal treatment of the data has been impossible (reflections are too close to be separated).

The structure was solved and further refined by the full-matrix least-squares method using the SHELXTL software package [23, 24]. The positional and anisotropic displacement parameters have been refined for all atoms using all the measured reflections in refinements on F^2 . The total filling of the atomic sites has been checked by free refinement of the occupational factors which did not deviate from unity within the standard deviation limits.

For both crystals, the recorded reflections have been indexed in a primitive hexagonal lattice ($a = b \approx$ 7, $c \approx 10$ Å) and the statistical tests on intensities indicated a non-centrosymmetric space group. But, because the structure of KSbGe₃O₉ was previously reported, from X-ray powder diffraction studies, with the larger hexagonal cell a = b = 13 Å and c = 10 Å in the $P\overline{6}c2$ space group [22], we also tried to index our data in a 2×2×1 supercell, which gives rise to additional reflections (h = 2n+1) of extremely low intensity and high $\sigma(I)$ values. No solution was found in this supercell and the refinement of a model built from twice the sub-cell structure did not converge either properly, giving arguments to reject a parameter doubling.

Furthermore, structure solutions have been found in the basic cell of parameters $a=b \approx 7$, $c \approx 10$ Å, and those obtained in the most probable space groups differ only by the *z*-coordinates of Ge and O atoms. Constrained by the symmetry in $P\overline{6}c2$ (z=0 or 0.25) or free to vary in P3c1, they lead to Ge₃O₃

rings either flat or wavy. Refinement of the structural model in the lowermost symmetry P3c1 (which allows the greatest freedom degree) conducted for both crystal data sets shows that the deviation, in regard to the plane, remains lower than 0.007Å within the Ge₃O₃ rings. In these conditions, the structure is given in the highest hexagonal symmetry $P\overline{6}c2$ usually reported for the AMGe₃O₉ compounds [10-15]. An essential characteristic in the structure of KSbGe₃O₉ lies in the flatness of the Ge₃O₃ rings, a situation differing from RbSbGe₃O₉ which a structural study has just been published [17].

The main crystal data and refinement parameters are given in Table 1 while the atomic positions and the equivalent displacement parameters are listed in Table 2. The full CIF files (CCDC numbers 1990049-50) can be obtained free of charge from the Cambridge Crystallographic Data Center [25].

Space group	$\overline{p}\overline{6}_{22}$	(188)	
Space group	P6c2(188)		
Temperature (K)	298 (2)		
Diffractometer, radiation	Bruker D8 Venture, Ka Mo		
Wavelength (Å)	0.7107		
Lattice (Å)	a = 6.9436(1), c = 9.9934(2)	a = 6.9527(1), c = 10.0117(2)	
Volume (Å ³)	419.13(2)	418.27(2)	
Z	2		
Density (g/cm ³)	4.14	4.15	
μ (mm ⁻¹)	14.4		
Absorption correction	multi-scan method (SADABS)		
Crystal (mm)	$0.086 \times 0.085 \times 0.042$	0.10×0.08×0.03	
θ range (°)	3.39 - 62.22	3.38 - 72.37	
Index ranges	-16≤h≤17, -17≤k≤17 -17≤h≤18, -18≤k≤18		
	-24 <u>≤</u> 1 <u>≤</u> 24	-26 <u>≤</u> 1 <u>≤</u> 26	
Collected reflections	40471	69478	
Independent reflections	2313 [$R_{int} = 0.0568$]	$2886 [R_{int} = 0.0522]$	
Goodness-of-fit on F ²	1.271	1.132	
Final R indices $[I > 2\sigma(I)]$	R1 = 0.0324, $wR2 = 0.0578$	R1 = 0.0233, wR2 = 0.0550	
R indices (all data)	R1 = 0.0377, wR2 = 0.0592	R1 = 0.0245, wR2 = 0.0552	
Absolute structure parameter	0.043(16)	0.083(11)	
Extinction coefficient	0.044(3)	0.040(3)	
$\Delta \rho$ Fourier residuals (e.Å ⁻³)	3.49 / -1.72	2.29 / -1.96	

Table 1. Crystal structure and refinement parameters for the two studied crystals of KSbGe₃O₉.

	Х	У	Z	U_{eq}
Sb	0	0	0	613(3)
	0	0	0	686(4)
Ge	0.12135(3)	0.37281(3)	0.25	706(1)
	0.12133(4)	0.37280(4)	0.25	784(4)
Κ	0.66667	0.33333	0	3850(5)
	0.66667	0.33333	0	3840(5)
01	0.0663(3)	0.5931(3)	0.25	1560(3)
	0.0668(3)	0.5935(3)	0.25	1620(4)
02	0.2411(3)	0.0136(4)	0.1071(2)	2690(3)
	0.2409(4)	0.1140(4)	0.1073(2)	2730(4)

Table 2. Atomic coordinates and equivalent displacement parameters ($Å^2 \times 10^5$) for KSbGe₃O₉ (*italics* for crystal 2). U_{eq} is defined as $\frac{1}{3}$ of the trace of the orthogonalized U_{ij} tensor.

The compound KSbGe₃O₉ is isostructural with the benitoite BaTiSi₃O₉ which is known as a structural type [10, 11, 15]. It is characterized by SbO₆ octahedra and GeO₄ tetrahedra which arrangement provides [Ge₃O₉]⁶⁻ units formed with Ge₃O₃ planar rings (see in Figure 2 the structure representations drawn using Diamond software [26]). The relevant interatomic distances and angles in this three-dimensional open-framework structure are given in Table 3. The non-distorted SbO₆ octahedra share its vertices with six GeO₄ tetrahedral units, the latter display regular tetrahedral geometry and share vertices with two GeO₄ tetrahedra and two SbO₆ octahedra. The SbO₆ octahedra are separated by Ge₃O₃ rings. The voids between the polyhedra, filled with K atoms, form channels along the *c*-axis. The atomic arrangement in KSbGe₃O₉ can also be viewed as composed of two planes of Ge₃O₃ rings perpendicular to the *c* axis (at z = 0.25 and z = 0.75) followed, above and below, by an oxygen layer and then a layer containing both Sb and K atoms.

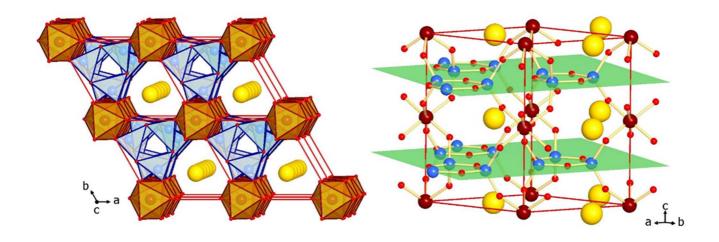


Figure 2. Structural representations of KSbGe₃O₉ emphasizing: *left*-the GeO₄ (blue) and SbO₆ (brown) regular polyhedral packing forming channels containing the K^+ ions (yellow). *right*-the Ge₃O₃ planar rings in the *ab* plane.

Sb-O2	1.0405(2) ~ (O1-Ge	1.741(2)
	$1.9495(2) \times 6$	O1-Ge	1.754(2)
K-O2	2.875(2) × 6	Ge-O2	1.7311(2) ×2
	$2.873(2) \times 0$	Ge-O1	1.741(2)
O2-Sb-O2	83.91(1) × 3	Ge-O1-Ge	129.56(1)
O2-Sb-O2	$175.04(2) \times 3$	O2-Ge-O2	111.10(2)
O2-Sb-O2	92.61(1) × 6	O2-Ge-O1	$114.28(9) \times 2$
O2-Sb-O2	91.08(2) ×3	O2-Ge-O1	$102.74(8) \times 2$
		01-Ge-O1	110.44(1)

Table 3. Selected bond lengths (Å) and angles (°) for KSbGe₃O₉

3.3. Vibrational properties

The zone-center optical phonons of KSbGe₃O₉ can be classified according to the irreducible representations of the D_{3h}-point group as $7A'_1 \oplus 9A'_2 \oplus 6A''_2 \oplus 5A''_1 \oplus 15E' \oplus 12E''$. The E'-representation is both infrared and Raman active while the E''-representation is only Raman active. The A''_2 (resp. A'_1) representation is infrared (resp. Raman) active. The A'_2 and A''_1 modes are silent. Due to the lack of an inversion center in the crystal structure, the macroscopic electric field splits the

polar modes (infrared modes) into transverse optical (TO) and longitudinal optical (LO) modes close to the center of the Brillouin zone.

The room-temperature infrared and Raman spectra of the non-centrosymmetric KSbGe₃O₉ material are presented in Figure 3.

In the 450 - 1100 cm⁻¹ frequency range, the hexagonal KSbGe₃O₉ material presents 5 main experimental IR bands in Figure 3a centered at 498, 570, 798, 841, and 920 cm⁻¹ and 4 shoulders at 851, 878, 943 and 1018 cm⁻¹ in agreement with the literature reports [27]. These 5 main experimental bands are supported by our DFT calculation as evidenced in Figure 3a. The experimental shoulders could be combination bands or defects as their counterparts are not predicted by our calculations.

A same observation can be made regarding the high-resolution Raman data, Figure 3b. Furthermore, the frequency positions of the main experimental Raman bands are also in very good agreement with the literature [27].

This very good agreement between the experimental vibrational spectra and the theoretical ones support our structure refinement for the non-centrosymmetric KSbGe₃O₉ material and points out the high quality of the as-grown crystals.

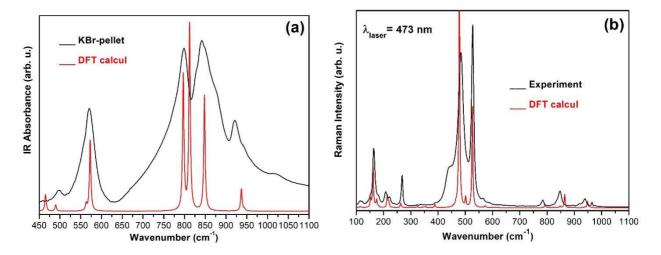


Figure 3: Room-temperature experimental and 0 K-calculated (a) infrared and (b) Raman spectra of the KSbGe₃O₉ polycrystalline sample.

The silent representations calculated at 0 K for the KSbGe₃O₉ crystal (D_{3h} point group, cell constants a = 6.9436(1) Å, c = 9.9934(2) Å) are given in Table 4. Both the experimental and DFT frequency positions along with their symmetry are presented in Table 5. All the calculated zone-center optical frequencies have been assigned to observed bands in the Raman and IR spectra.

Symmetry	Calculated wavenumbers (cm ⁻¹)	
A'2	74	
A'2	145	
A"1	156	
A'2	213	
A"1	240	
A'2	251	
A"1	350	
A'2	408	
A'2	473	
A"1	495	
A'2	527	
A'2	858	
A'2	915	
A"1	965	

Table 4. A'₂ and A"₁ silent mode position for KSbGe₃O₉ in the D_{3h} point group.

Table 5. Position of the experimental and calculated optical modes (infrared and Raman) with theirMulliken symmetry for KSbGe $_{3}O_{9}$ in the D $_{3h}$ point group.

Calculated wavenumbers (cm ⁻¹)	Symmetry	Observed wavenumbers (cm ⁻¹)		
		Raman Expt.	Infrared Expt.	
65	E" (TO)			
68	A" ₂ (TO)			
92	E" (TO)			
111	E' (TO)	108		
114	E' (TO)	113		
115	E" (TO)	118		
133	A" ₂ (TO)			
153	E" (TO)	158		
161	A' ₁ (TO)	164		
170	E' (TO)	168		
175	E" (TO)	174		
181	E" (TO)	183		
213	E' (TO)	207		
220	E' (LO)	222		
233	E' (TO)	233		
259	A" ₂ (TO)			
263	A'1 (TO)	268		
267	E' (TO)	278		
285	E" (TO)	300		
324	E' (TO)	323		
326	A" ₂ (TO)			
345	E" (TO)			
388	A'1 (TO)	392		
427	E' (TO)	439		
457	E" (TO)			
465	E' (TO)	464	498	
478	A'1 (TO)	485		
490	A" ₂ (TO)		520	
501	E" (TO)	495		
526	A' ₁ (TO)	527		

563	E' (TO)	562	550
572	E' (TO)	566	570
792	E" (TO)	787	
797	E' (TO)	795	798
812	A" ₂ (TO)		798
848	E' (TO)	838	841
863	A'1 (TO)	849	
864	E" (TO)	849	
901	E' (LO)		878
937	E' (TO)	913	920
944	E' (TO)	919	943
948	A' ₁ (TO)	940	

3.4. Vibrational frequency assignments.

The structural origin of the vibrational components was interpreted using the results of the density functional theory calculations. Only the most intense IR and Raman vibrational bands were investigated. The theoretical vibrations in the range $450 - 1000 \text{ cm}^{-1}$ are mainly due to the oxygen motions and, for some of them, to the vibrations of the Ge atoms in a small content. Within the Ge₃O₉ cyclic unit, there are two types of oxygen atoms (see Table 2): the bridging O (O1) bonded to two Ge atoms (Ge-O1-Ge forming the Ge₃O₃ planar ring) and the non-bridging O (O2) bonded to one Ge atom (Ge-O-K/Sb).

The first intense Raman line is found at 478 cm⁻¹ by DFT calculation against 485 cm⁻¹ experimentally, Figure 3b and Table 5. This mode is assigned as A'₁ and it is only Raman active. This vibration is mainly caused by the symmetric stretching of O2-K "pairs", which leads to the bending of the O2-Sb-O2 angle. In this mode, the K atoms are fixed and the Ge atoms too, Figure 4.

The 563 cm⁻¹ calculated frequency which is both IR and Raman active (E'(TO), Table 5) is due to Ge-O1-Ge intra-cycle bending motions in the Ge₃O₃ ring and to the O2-K-O2 wagging motion of the KO₆ octahedra, Figure 4. Both O and Ge atoms participate in this vibration mode while K and Sb atoms are fixed. This band characterizing the annular units appear in the IR/Raman spectra at a frequency of 500-570 cm⁻¹ in three-membered rings germanates and is called the "ring band" [27, 28, 29].

Concerning the 797 cm⁻¹ E'(TO) vibration frequency, it concerns the symmetric stretching in the *ab* plane of the Ge-O1 bonds and the twisting motion of O2-K-O2 bridges, Figure 4. Both K and Sb atoms are fixed.

The O1-Ge-O1 wagging of the Ge₃O₃ central cycle along the *c* axis and the O2-K-O2 wagging of the KO₆ octahedra occur in the $A^{"}_{2}(TO)$ band at 812 cm⁻¹ which is only IR active, Table 5 and Figure 4. For this mode, the dominant motions are due to both the Ge atoms forming the Ge₃O₃ rings and the O2 atoms. The displacement of the Sb atoms is small while that of K is null.

The 848 cm⁻¹ E'(TO) theoretical vibration, Table 5, is due to intra-cycle symmetric stretching motions of the O1 atoms concomitant to the shear in the *ab* plane of the ring Ge_3O_3 and to the O2-K-O2 twisting, Figure 4. The Sb and K atoms do not move.

The 937 cm⁻¹ calculated frequency which is both IR and Raman active (E'(TO), Table 5) is attributed to the Ge-O1 symmetric stretching in the Ge_3O_3 ring and to the vibration associated with the rotation of whole KO₆ octahedra (libration mode), Figure 4. The Sb and K atoms do not move.

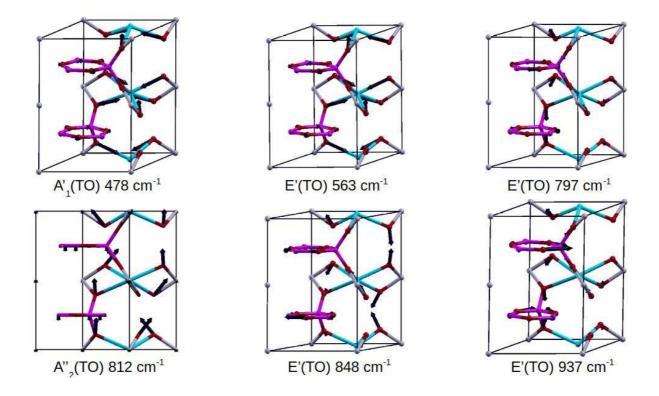


Figure 4. Structural origin of some vibrations in the non-centrosymmetric KSbGe₃O₉ compound

4. CONCLUSION

The structural and vibrational properties of the hexagonal KSbGe₃O₉ crystal have been studied for the first time by using experimental data combined with first-principles based calculations. The main conclusions are the following:

(1) This non-centrosymmetric phase ($P\overline{6}c2$) is visually colorless and transparent, and it is structurally stable from room-temperature up to, at least, 1200°C.

(2) The unit cell parameters refined from X-ray single-crystal data are different from those given in JCPDS card number 00-038-1334.

(3) KSbGe₃O₉, with planar Ge₃O₃ rings, belongs to the benitoite-type structure.

(4) The observed IR and Raman band positions are in good agreement with the calculated ones.

(5) The structural origin of the most intense vibrations is mainly due to the oxygen motions.

(6) Two E'(TO) vibrations (IR and Raman active) characterize the planarity of the Ge₃O₃ ring in the *ab* plane.

This anisotropic oxide with a stoichiometric chemical composition and a large thermal structure stability may have potential piezo-electrical and non-linear optical properties exploitable in a range of very high temperatures up to, at least, 1200°C. Lead-free materials that can operate under extreme stress without degradation are sought for industrial needs. Furthermore, this germanate with its open-framework structure and well-defined channels may be suitable for ion-exchange and gas adsorption applications.

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