Impact of polyelectrolyte multilayers on the ionic current rectification of conical nanopores.

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Abstract :

Keywords :

1. Introduction

For the last two decades, single nanopore technology has contributed to the emergence and design of nano-fluidic responsive systems [1, 2]. It has provided a deep understanding of the transport from simple electrolyte to (bio)macromolecule (DNA, polymer or protein) at the nano-scale, contributed to the development of new analytical device for DNA sequencing. Nanopores with a dissymmetry of charge or shape are also particularly fascinating for its ionic diode [3-5]. The latter are usually obtained using conical or semi-cigare shape nanopores in polymer film such as PET, PC or PI using track-etched technique [6]. Ionic diode behavior comes from a dissymmetrical distribution of the ions inside the

nanopore which induces a depletion of anions or cations [7, 8]. It can be tuned by the functionalization of the surface of the nanopore walls. For PET, the carboxyl moieties are particularly interesting because they can be easily functionalized with a large range of molecules or macromolecules [9-11]. Many stimuli responsive channels such as pH [12, 13], light [14, 15] have been designed using this strategy. As the current rectification is strongly dependent of the nanopore surface charge, it can be used to develop sensors, when molecules or macromolecules are attached at its surface [16, 17].

Amongst all the possibilities offered by track-etched nanopores, the ability to modify the ionic transport with pH is particularly interesting. It involves the grafting of functions onto the nanopore surface wall, exhibiting different charges as a function of pH. For example, poly-protonic polymer brushes permit a modulation of the current rectification [18]. Amphoteric molecules such as lysine or histidine [19] were used to modify the nanopore selectivity by changing the pH. At low pH, the nanopore is positively charged and selective to anion, while at high pH it becomes negatively charged and thus selective to cation. Similar properties can be obtained using poly(methacryoyl-L-lysine) brushes. Spiropyrans modify the ionic transport by modifying either or both pH and light [14]. All those strategies are based on the chemical grafting of responsive molecules onto the carboxyl moieties.

The layer-by-layer deposition of polyelectrolyte is an interesting method to functionalize membranes and nanopores to obtain ionic diode behaviors. It allows to improve their ionic selectivity, and reduces the fouling by proteins [20, 21]. The deposition of poly-L-lysine can inverse the current rectification. The advantage of this functionalization is its reversibility. Due to the electrostatic interactions of the electrolyte with the surface, reversible and modular sensors can be designed. Poly(allylamine hydrochloride) (PAH) poly(styrenesulfonate) (PSS) have been used to solve by a mathematical model the exact shape of nanopore [22]. Based on finite element simulation of PNP equation, the authors demonstrated that the surface charge of the pore walls decreases dramatically with the number of PAH/PSS layers assembled into the nanopore, the current rectification tends to 1.

We assume that confined multilayers of polyelectrolytes create a zwitterionic film onto the nanopore surface. It should exhibit similar properties to the ones observed after grafting amphoteric molecules. We expect to obtain a nanopore selective to anion at low pH and cation at high pH. Such behavior had been reported for a conical nanopore functionalized with chitosan/ polyacrylic acid (PAA) [23]. However, the behavior of polyelectrolytes is complex and strongly depends on the pH, especially for the weak ones by changing ionic strength or pH induces swelling of the structure of the polyelectrolyte layer as well as modifying the ionic transport. This was evidenced by nanopores functionalized with polyethylenimine (PEI) and chondroitin-4-sulfate (ChS) layers[13]. The swelling of ChS which occurs at low pH induces original gating properties. The nanopore closes at $\text{pH} > 4$ and let only the anions go through the nanopore at $\text{pH} < 4$. This phenomenon only occurs at low salt concentration. Regarding the opposite behavior of these two systems, it is necessary to go further in the investigation on the ionic transport properties of the conical nanopore functionalized by polyelectrolyte multi-layer.

In this work, we have functionalized single conical nanopore in PET, obtained by track-etched technique with different couples of polyelectrolytes: poly-L-lysine (PLL) /PAA and PEI/PAA. The influence of pH and divalent cation is reported. Spectroscopic ellipsometry experiments were performed on surfaces to know the influence of salt and pH on the thickness of the multilayers of polyelectrolytes. Finally, we also reported the impact of cross linking of several polyelectrolyte layers in order to prevent large conformational changes.

2. Material and Methods

2.1. Material

Poly(ethylene terephthalate) (PET) film (thickness 13 μm , biaxial orientation) was purchased from Goodfellow (ES301061). Sodium chloride (71380), potassium chloride (P3911), sodium hydroxide (30603), hydrogen chloride (30721), branched PEI (Mw 25 kD, 408727), PAA (Mw 100 kD, 523925), PLL (30 kD-60kD P2636) were purchased from Sigma-Aldrich. Ultra-pure water was produced from a Q-grad®-1 Milli-Q system (Millipore).

2.2. Track-etching nanopores and characterization

Single tracks were produced by Xe irradiation (8.98 MeV) in PET films (13 μm) (GANIL, SEM line, Caen, France). The PET films were exposed to UV light for 24h per side to activate the track (Fisher bioblock; VL215.MC, $\lambda = 302 \text{ nm}$), before the chemical etching process. The etching of the conical nanopore was performed under dissymmetric condition following the electro-stopping method.[5] The PET foil was mounted between the two compartments of a chemical cell in Teflon. The etchant solution (NaOH 9 M, 1.6 ml) was added on the base side and the stopping solution (KCl 1M and acetic acid 1M, 1.6 ml) on the tip side. A potential of 1 V was applied across the membrane; the reference electrode is immersed in the stop solution and the working one in the etchant solution. When the current reaches a value of several hundred pA, the etching process was stopped by the replacement of the etching solution by the stop one.

The conical nanopores are characterized as follow. The base diameter (D) is calculated as a function of the total etch time t (min) using the relationship determined in our lab $D = 2.5t$. -The tip diameter (d) was calculated, assuming a bulk-like ionic conductivity inside the nanopores, from the dependence of the conductance G to the NaCl concentration (equation 1).[24] In order to correct the conductivity at high salt concentration, the diameter is calculated using the ionic conductivity of solution κ . The latter have been measured using a conductimeter (Hanna HI 255 combined meter with conductivity and electrode HI 76310) after preparation.

$$G = \kappa \pi D d / L \quad (1)$$

where L is the nanopore length.

2.3. Current-voltage measurements

Electrical measurements were performed using a patch-clamp amplifier (EPC10 HEKA electronics, Germany). The current is measured by Ag/AgCl electrode. A single nanopore was mounted in the

chemical Teflon cell containing an electrolyte solution. One electrode was plugged to the working end of the amplifier (*trans* chamber, base side) and the other electrode connected to the ground (*cis* chamber, tip side). Recorded currents were analyzed by Fitmaster (Heka Elektronik, Germany).

For I–V curves, the currents data were recorded as a function of the time under constant voltage from -1 V to 1 V by 100 mV step and from -100 mV to 100 mV by 10 mV steps. All current traces were recorded during 10 s at a frequency of 50 kHz. These measurements were performed 3 times. The conductance G is extracted from the linear zone of I-V curves from typically - 75 mV to 75 mV.

2.4. Nanopore functionalization

PAA, PEI and PLL stock solutions at concentration of 1 mg ml⁻¹, pH 7 were prepared. The functionalization of the nanopores was performed under KCl 0.1 M pH 7 by addition of 16 µl of stock solution at the tip side. -1V was applied for PEI and PLL during 5 minutes. Then the cell was filled with buffer to remove the excess of polyelectrolytes. Then 16 µl of PAA solution was added at the tip side during 5 minutes applying 1V. The procedure is repeated until the desired numbers of PEI/PAA or PLL/PAA bilayers. During the deposition, the current is recorded at a 5 kHz frequency in order to monitor the adsorption of the polyelectrolytes.

Table 1. list of the nanopores used in this work

Nanopore	d	D	(Polyelectrolyte) _{#layers}		Code
NP-1	30	300	(PLL) ₃ /(PAA) ₃		31PET13B20
NP-2	28	475	(PEI) ₄ /(PAA) ₄		29PET13 A12
NP-3	116	346	(PLL) ₁₀ /(PAA) ₁₀		30PET13C02

NP-4	91	460	(PEI) ₈ /(PAA) ₈		30PET13C04
NP-5			(PLL) ₄ /(PAA) ₄	crosslinked	

2.5. Spectroscopic ellipsometry

Ellipsometric investigation of PLL/PAA and PEI/PAA bilayers were performed using spectral ellipsometer from J. A. Woollam M2000X on silicon wafer substrate. Firstly surface of the wafer was activated using piranha solution (H₂SO₄:H₂O₂ 3:1) to form -OH groups at the surface, necessary for the immobilization of the polyelectrolyte layer. Ellipsometric spectra were recorded from 200 nm to 1000 nm, angle of incident light was 70°. Cauchy dispersion function was applied for the regression analysis of the formed PEI, PAA and PLL layers. Optical constants and thicknesses of these layers were obtained using Complete EASE software.

3. Results and discussion

3.1. Polyelectrolyte deposition.

Single conical nanopores were obtained in PET film (thickness 13μm) by track-etched method. The nanopores used for this study are reported in Table 1. More than 8 bilayers can be deposited into conical nanopores with a large tip diameter ($d \sim 100$ nm). For small tip diameter ($d \sim 30$ nm) less than 4 bilayers was used. The nanopore internal surface is covered by carboxylate groups and thus exhibits a negative surface charge at pH 8. The I-V dependence recorded from -1 V to 1 V curves show a current rectification characterized by the rectification factor (f_{rec}) calculated from $f_{rec} = |I_{(1V)}|/|I_{(-1V)}|$. Under our experimental condition (Figure 1), the current rectification is lower than 1 when the nanopore is negatively charged. The polycation (PEI or PLL) or polyanion (PAA) deposition was performed by adding them at the tip side under a voltage of -1V or 1V respectively. This voltage provides the required energy at the polyelectrolyte to overcome the barrier of entrance through nanopore. Independently of the

polyelectrolyte nature, the current recorded as a function of the time shows a strong decrease, which is induced by the charge inversion at the nanopore entrance [21]. Indeed after the adsorption of the polycations, the nanopore becomes positively charged and thus the rectification factor increase above 1 (Figure 1), while after the adsorption of the polyanions, the nanopore becomes negatively charged and the rectification factor decrease below 1. The inversion of the rectification factor brings evidence that each layer of polyelectrolytes has been deposited successfully. According to previous work, we can assume that the polyelectrolyte deposition occurs at the narrow opening.

The dependence of f_{rec} with the number of bilayer strongly depends on the polyelectrolyte couple. For (PLL)₁₀/(PAA)₁₀, it tends to a value around 10⁻³. When increasing the number of bilayers, the addition of PLL does not induce a clear inversion of the rectification factor. Similar behavior was previously reported in the case of PEI/ChS deposition [13]. It was assigned to a partial charge compensation between either polyelectrolytes or an in-homogeneous surface charge patterns between the region close to the narrow opening and the rest of the nanopore. For PEI/PAA, the inversion of rectification occurs after each deposition likely due to a better charge compensation between both polyelectrolytes. However, it is interesting to note that f_{rec} does not tend to 1 as reported for PAH/PSS.

Figure 1 : characterization of the nanopore functionalization : I-V dependence after polyelectrolyte deposition recorded at NaCl concentration of 0.1 M (a) NP-3 (b) NP-4. Rectification factor after the deposition of each electrolyte (c) NP-3 (d) NP-4. Time required to observe the inversion of current rectification induced by the polyelectrolyte deposition (e) NP-3 (f) NP-4

The time required for the polyelectrolyte to be deposited is very dependent on the nature of the polycation. In the case of PLL, PAA is deposited in less than 120 s for the 7 first bilayers and requires more time for the layers 8 to 10 probably due to steric hindrance. With PEI, the deposition of PAA

requires more than 2000 s. The difference of time is clearly dependent of the polycation which can influence the structure of the bilayer inside the nanopore. In order to characterize this behavior, the thickness of PLL/PAA and PEI/PAA bilayers were investigated using spectroscopic ellipsometry. The polyelectrolyte bilayers were deposited on the surface of a wafer. The obtained values of refraction index dispersion are presented in Figure 2 and thicknesses of the layers are presented in **Erreur ! Source du renvoi introuvable.** Figure SI-1 of raw Psi and Delta data, shows that additional bilayers are formed on the top of the silicon substrate. The building of an optical model becomes very complicated in this case when more bilayers are deposited on the substrate. Optical model well fits data for single PLL, PEI layers and for PLL/PAA and PEI/PAA bilayers only. After fitting using Cauchy dispersion formula, refractive index dispersion and thicknesses of these layers are calculated. Spectroscopic ellipsometry characterization evidences a difference of bilayer thickness which are 7.28 nm and 3.91 nm for PEI/PAA and PLL/PAA respectively. This difference is mainly due to the thickness of the PAA layer and thus its conformation which is more folded in presence of PEI. The longer time can thus be interpreted by steric effect which leads to a more complicated penetration of the polyelectrolyte inside the nanopore and/or the necessity to add more polymer chains to cover the polycation layer.

Figure 2 : Refractive indexes dispersion obtained for PAA, PEI and PLL layers deposited on silicon wafer substrate.

Table 2. Thicknesses of PLL, PAA, PEI layers on silicon wafer

	PEI/PAA		PLL/PAA	
	PEI	PAA	PLL	PAA
thickness, nm	2.14	5.14	1.68	2.23

3.2. Ionic transport through nanopores modified by LbL deposition

The ionic transport properties of the nanopore decorated with the polyelectrolyte were studied for different NaCl concentrations (from 10^{-3} until 1 M) at various pHs (Figure 3). Because the pKa of PAA is 4.5, we can expect that at pH 3 the nanopore will be globally positively charged inducing an inversion of the rectification factor.

3.2.1. PEI/PAA LbL functionalization of a nanopore

In Figure 3 are reported the I-V dependence at pH 3 and 7 for two nanopores with large and small diameters. For the large nanopore ($d = 91$ nm) functionalized with 8 layers of PEI/PAA, the I-V dependence shows a rectification factor below 1 at pH 7. Interestingly, at pH 3, the inversion of the rectification factor occurs at low salt concentration ($\text{NaCl} = 0.01$ and 0.05 M). When increasing the concentration, the nanopore seems closed and the rectification factor is close to 1. pH-gating properties under acidic conditions were previously reported for nanopore decorated with PEI/ChS [13]. However in that case, with the increase of the salt concentration, the gating property disappeared. With the PEI/PAA, the interesting result is that the nanopore is selective to anion at low salt concentration at pH 3. To confirm this property, the same study was performed in a conical nanopore with a smaller tip diameter (29 nm) decorated with 4 PEI/PAA bilayers. The I-V dependences reported on Figure 3 reveal the same behavior. For these two nanopores, the enclosure at pH 3 arises for a NaCl concentration of 0.1 M. At pH 4, the nanopore seems closed independently of the NaCl concentration.

NP-2

NP-4

Figure 3 : I-V dependance of NP functionalized with PEI/PAA LbL at pH 7 and pH 3

To go further in our investigations, we studied the evolution of the nanopore conductance as a function of the NaCl concentration for different pHs (Figure 4). Inside a charged nanopore, the ionic conductance decreases with the salt concentration until a plateau. This constant conductivity is due to the mobile counter ions which shield the surface charge and the slippage [25-27]. At pH upper than 4, the conductance decreases with the salt concentration without reaching clearly a plateau. This result suggests a global compensation of charges between the polyations and polyanions. It is also consistent with the

charge inversion observed after each polyelectrolyte deposition. At pH 3, the conductance exhibits an unusual dependence with the salt concentration. At concentration below 0.1 M, the conductance decreases with the salt concentration. In addition, the value of conductance is higher compared to the one obtained at greater pH. This can be explained by the disequilibrium of charge induced by the p protonation of the carboxylate groups. In order to counterbalance the excess of positive charges, Cl^- ions are “pumped” inside the nanopore, increasing the concentration of mobile ions inside the nanopore. For NaCl concentration of 0.1 M, the conductance decreases. This nanopore enclosure could be assigned to the reorganization of the polyelectrolyte layers.

Figure 4 : dependence of the conductance vs the NaCl concentration of NP functionalized with PEI/PAA LbL from pH 7 to pH 3

3.2.2. Conical nanopores decorated with PLL/PAA LbL

The NaCl transport through nanopore functionalized with PLL/PAA LbL exhibits completely different behavior (Figure 5). The rectification is lower than 1 at pH 7. The conductance as function of the NaCl concentration reveals a plateau at low salt concentration as usually reported in the case of charged nanopore. The constant conductance is attributed to a partial compensation of charge between PLL and PAA. This is consistent with the fact that the PLL deposition cannot allow an inversion of rectification factor. The value of the conductance plateau is higher at pH 4 and 3. This can be explained by the protonation of PAA which breaks the partial balance of charge between both polyelectrolytes. In this

case, the global charge of the polyelectrolyte layers becomes globally positive and thus we could expect an inversion of rectification factor. The I-V curves recorded at pH 3 for two nanopores with different diameters and number of layers exhibit an opposite behavior. Indeed, at pH 3 the rectification factor is below 1. This can be interpreted using two assumptions. The first one is the modification of PAA pKa in nanopore as predicted by theoretical model [28] and supported by previous work[29]. The second one is a modification of the polyelectrolyte conformation as it occurs for PEI/PAA or PEI/ChS [ref].

Figure 5 : I-V dependence (a-b), rectification factor (c) and conductance as a function of NaCl concentration (d) of NP functionalized with PEI/PAA LbL at pH 7 and pH 3.

In order to investigate how pH affects the formed bilayers, we studied the impact of pH on PLL/PAA bilayers by total internal reflection spectroscopic ellipsometry (TIRE) (Figure 6). When increasing the NaCl concentration from 0.01 M to 1 M, shift to the longer wavelength indicates a swelling of the

bilayers. At 0.01 M and 1 M of NaCl curves are near each other at pH 3 and 7, the pH does not influence the bilayer swelling. Usually, weak polyelectrolyte layer like PAA swells at pH below their pKa [30]. According to the TIRE results, the non-inversion of the I-V dependence at pH 3 could be attributed to a displacement of the pKa of PAA.

Figure 6 : Real time monitoring of ellipsometric parameters Psi and Delta using pH and NaCl concentration

3.3. Ionic transport through nanopores functionalized by crosslinked LbL deposition

The simple deposition of polyelectrolyte bilayer shows interesting properties as pH gated in the case of PEI/PAA. However, the swelling and the displacement of pKa does not make the nanopores selective to anion at low pH and to cation at high pH. As previously mentioned, this goal was previously reached with chitosan/PAA. The main difference with the polycation used in this work is that chitosan is not soluble at pH upper than 5 and thus the bilayer cannot swell. In order to stabilize the polyelectrolyte layer, one solution is to crosslink the amine and carboxylic moieties after deposition. Using this strategy, we functionalized conical nanopore with PLL/PAA, after their deposition, was added at pH 8 to ensure the crosslinking. The evidence of each step was characterized by a modification of the rectification ratio. After functionalization, the I-V response was recorded at different NaCl concentrations from 10^{-3} M to 1 M at pH 8 and 3 (Figure 7). They follow the same behavior independently of the NaCl concentration. At

pH 8, the current rectification is below 1. This means that the nanopore enhances the transport of cation (here Na^+) compared to Cl^- . This is confirmed by the permeation ratio $P_{\text{Na}^+}/P_{\text{Cl}^-} = 8$ measured under dissymmetrical condition. This result is assigned to a charge globally negative of the nanopore. This is a reasonable assumption because the last layer is polyanions, the nanopore surface is totally deprotonated. At pH 3, an inversion of the current rectification occurs since the factor is upper than 1. In this case, the transport of Cl^- is favored compared to the Na^+ as confirmed by a permeation ratio $P_{\text{Na}^+}/P_{\text{Cl}^-} = 0.01$. This result is assigned to a global positive charge of the region close to the narrow opening where are located the polyelectrolyte layers. Indeed at this pH, the PAA nanopore surface are protonated and PLL are totally protonated.

Figure 7 : I-V dependence (a-b), rectification factor (c) and conductance as a function of NaCl concentration (d) of NP functionalized with PEI/PAA LbL at pH 7 and pH 3.

The conductance dependence with the salt concentration does not follow the same behavior at pH 3 and pH 8. At pH 3, the conductance decreases with the salt concentration until reaching a plateau. The constant conductance is assigned to the counter mobile anions present inside the nanopore to shield the excess of positive charges in the polyelectrolyte layers. At pH 8, the constant value at low concentration is not observed suggesting charge compensation between the electrolyte and eventually counter ion trapped inside layer. This interpretation seems in contradiction with the rectification below 1 suggesting a preferential transport of cation. But it has to be stressed out that the functionalization occurs only at the narrow aperture, the rest of the nanopore is negatively charged. This dissymmetry of charges along the nanopore explains the current rectification.

4. Conclusion

To sum up, our work aimed to fashion a nanopore which offers the possibility to drive the ionic properties by pH. To do so, we have explored an alternative method compared to the chemical grafting of amphoteric molecule. The starting assumption was that polyelectrolyte multilayer create a zwitterionic film on nanopore surface where the charged can be modulate by the pH. The simple deposition of PLL/PAA and PEI/PAA LbL cannot allow to validate this assumption due to the swelling of the polymer layer. However, interesting ionic transport properties were obtained for PEI/PAA because at pH 3 the nanopore is only at low salt concentration while at pH 7 it is already open. In order to prevent the swelling, the crosslinking of PLL/PAA was performed. As assumed in this case, the pH allows driving the ionic diode properties. Because, the track-etched technique allows to obtain a single pore as well as multipore, the upscale of these properties to a membrane can be considered. Beside the interest of this study to design responsive membrane, the work on a single nanopore allows a characterization of the behavior of confined polyelectrolyte layer.

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Supporting information: Design of pH-responsive nanopore using polyelectrolyte confinement.

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SI-1 Characterization of polyelectrolyte growth by spectroscopic ellipsometry

Figure SI-1a. a and b ellipsometric parameters Psi and Delta dependence on wavelength respectively for each additionally grown PLL/PAA layer, c and d ellipsometric parameters Psi and Delta dependence on wavelength respectively for each additionally grown PEI/PAA on the silicon substrate surface.

In order to investigate the kinetics of PLL/PAA layers formation, the total internal reflection spectroscopic ellipsometry was used. For this purpose custom made liquid cell for in-situ kinetics investigation was used together with attached 70 ° BK7 glass prism coated with a gold plate for surface plasmon resonance excitation in order to have a better sensitivity [In situ study of ligand–receptor interaction by total internal reflection ellipsometry, Z Balevicius, I Baleviciute, S Tumenas, L Tamosaitis, A Stirke, A Makaravičiute, A Ramanavičienė, A Ramanavicius, 2014/11/28, Thin Solid Films, 571, 744-748]. 10 PLL/PAA bilayers were grown onto the gold surface. After each additional layer was grown the cell was washed with deionized water. The shift of Psi and Delta curves after each washing step are presented in figure XXX a, b, c, d . The shift to longer wavelength at different time moments indicate that each additional layer is formed on the top of previously formed layer and after washing new layer remains on the surface.

Figure SI-1b. Real time monitoring of ellipsometric parameters Psi and Delta for PLL (a, b) and for PAA (c, d) layers on gold plate.

Figure SI-2 : I-V dependence of nanopore (d=98 nm) functionalized with PEI/PAA 10 bilayers

Figure SI-3 : rectification factor of nanopore (d=98 nm) functionalized with PEI/PAA 10 bilayers

Figure SI-4 : I-V dependence of nanopore (d=28 nm) functionalized with PEI/PAA 4 bilayers

Figure SI-5 : I-V dependence of nanopore ($d=28$ nm) functionalized with PLL/PAA 4 bilayers